



## Ultra-low threshold InAs/GaAs quantum dot microdisk lasers on planar on-axis Si (001) substrates

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Monolithic integration of efficient III–V light-emitting sources on planar on-axis Si (001) has been recognized as an enabling technology for realizing Si-based photonic integrated circuits (PICs). The field of microdisk lasers employing quantum dot (QD) materials is gaining significant momentum because it allows massive-scalable, streamlined fabrication of Si-based PICs to be made cost effectively. Here, we present InAs/GaAs QD microdisk lasers monolithically grown on on-axis Si (001) substrate with an ultra-low lasing threshold at room temperature under continuous-wave optical pumping. The lasing characteristics of microdisk lasers with small diameter ( $D$ ) around 2  $\mu\text{m}$  and sub-wavelength scale ( $D \sim 1.1 \mu\text{m}$ ) are demonstrated, with a lasing threshold as low as  $\sim 3 \mu\text{W}$ . The promising lasing characteristics of the microdisk lasers with ultra-low power consumption and small footprint represent a major advance towards large-scale, low-cost integration of laser sources on the Si platform. © 2019 Optical Society of America under the terms of the OSA Open Access Publishing Agreement

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### 1. INTRODUCTION

The advanced technology of silicon photonics has emerged as a promising candidate for next-generation chip-scale data-communication networks and data centers [1–4]. Due to the indirect band gap property of bulk group-IV materials, the integration of well-established III–V photonic devices on silicon is a promising solution to realize efficient Si-based light-emitting sources. Although the method of heterogeneous integration has been widely studied and developed [5,6], the method of monolithic integration is still preferred for low-cost and dense photonic integrated circuits (PICs) [7], which is challenged by the conjunction of large thermal, lattice, and polarity mismatches between the Si substrate and III–V layer [8,9]. Studies on the optimization of the III–V buffer layer have been performed [10,11], which have led to the demonstration of high-performance InAs/GaAs quantum dot (QD) lasers epitaxially grown on different types of intermediate buffer layers (GaAs, GaP, and Ge) and Si substrates, including off-cut ( $4^\circ$ – $6^\circ$ ) Si [12–16], patterned on-axis Si (001) [17–24], Ge-on-Si [25–30], and GaP/Si (001) substrates [31,32]. Very recently, we demonstrated InAs/GaAs QD ridge-waveguide lasers monolithically grown on CMOS-compatible, on-axis

Si (001) substrate with only III–As buffer layers [33], which represents a major advance towards the commercial success of Si-based photonic-electronic integration. In order to interconnect the active region of laser devices and Si-based passive components, the thick buffer layer needs to be overcome. The interconnection between these parallel planes can be achieved by using a photonic wire bonding method [34] or introducing a grating coupler for vertical light coupling in the Si plane [35]. In addition, other possible solutions have been demonstrated, such as using a hybrid integration method by combining the bonding method and monolithic growth [36] and applying epitaxial lateral overgrowth technology [37].

Compared with ridge-waveguide lasers or distributed feedback lasers monolithically grown on Si, whispering-gallery-mode (WGM) microdisk lasers with small footprint configurations and ultra-low threshold allow for incorporating compact and efficient laser sources on a CMOS-compatible platform. Recently, Wan *et al.* reported continuous-wave (CW) optically pumped QD microdisk lasers under 10 K with a threshold of 35  $\mu\text{W}$  [19] and CW lasing emission under room temperature with a lasing threshold of 200  $\mu\text{W}$  [21,38], in which the five stacked

InAs QD active layers were directly grown on a GaAs-on-V-grooved-Si template with emission around 1.3  $\mu\text{m}$ . *Li et al.* presented CW optically pumped microdisk lasers with five layers of QDs as gain material grown on exact Si (001) substrate, and a lasing threshold of  $\sim 652 \mu\text{W}$  was obtained at room temperature [20]. *Shi et al.* demonstrated a 1.55  $\mu\text{m}$  QD microdisk laser grown on an on-axis Si (001) substrate with a threshold of 1.6 mW at liquid-helium temperature (4.5 K) [39], and an ultra-low threshold of  $2.73 \mu\text{W}$  was obtained under pulsed pumping conditions at room temperature [40]. However, room-temperature CW-pumped microdisk QD lasers with an ultra-low lasing threshold directly grown on planar on-axis Si (001) substrate have not been reported.

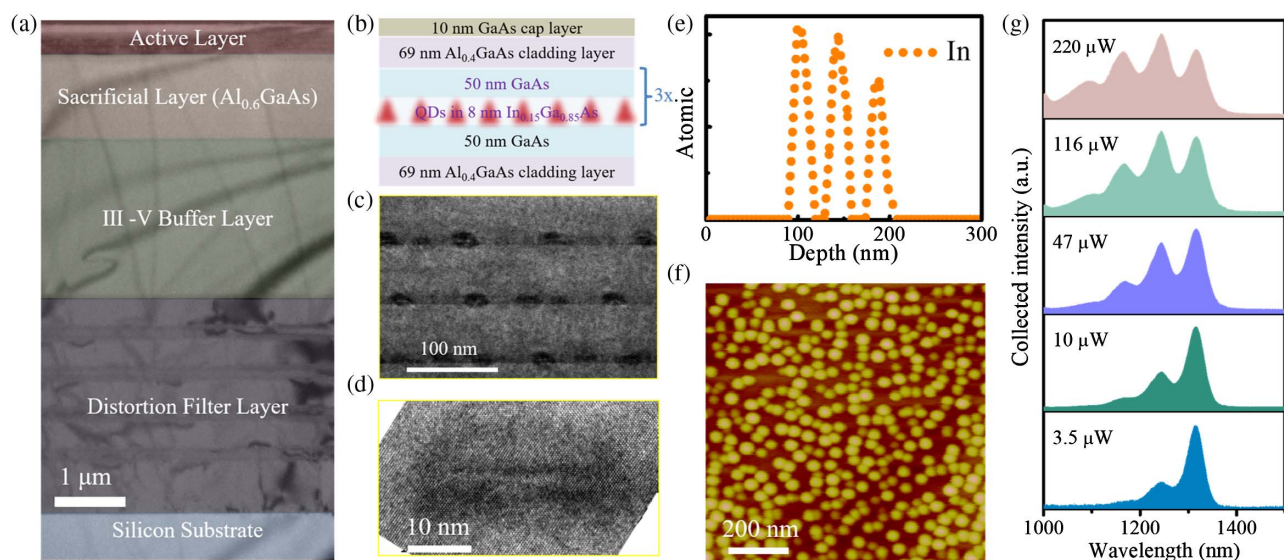
In this work, we demonstrate extremely low-threshold lasing in three stacked InAs/GaAs QD microdisk lasers monolithically grown on on-axis Si (001) substrate at room temperature by CW optical pumping. The lasing emission of microdisk lasers with a diameter ( $D$ ) of  $\sim 2 \mu\text{m}$  and a sub-wavelength scale  $D \sim 1.1 \mu\text{m}$  is illustrated, of which an ultra-low threshold  $\sim 3 \mu\text{W}$  is obtained. Lasing emissions from both the ground state and excited states are observed. The promising lasing characteristics of the microdisk lasers with an ultra-low lasing threshold and small footprint provide a viable route towards large-scale, low-cost integration of laser sources on the Si platform.

## 2. MATERIAL GROWTH AND FABRICATION

The InAs/GaAs QD microdisk lasers were monolithically grown on planar on-axis Si (001) substrates without any intermediate buffer layer [41]. A 400 nm epitaxial GaAs film was first deposited on the on-axis Si (001) substrate without antiphase boundaries using metal-organic chemical vapor deposition. A detailed description of the epitaxial structure between the active region and the Si (001) substrate, as well as the process of crystal growth, can be found in Refs. [33] and [41]. Figure 1(a) shows a cross-sectional transmission electron microscope (TEM) image of the microdisk structure grown on a planar on-axis Si (001) substrate.

The epitaxial structure of the microdisk region is schematically demonstrated in Fig. 1(b). Three stacked well-developed InAs/In<sub>0.15</sub>Ga<sub>0.85</sub>As/GaAs dot-in-well (DWELL) active layers were separated by a 50 nm GaAs spacer layer and sandwiched by two symmetrical 69 nm Al<sub>0.4</sub>Ga<sub>0.6</sub>As cladding layers capped by a 10 nm GaAs layer. Figures 1(c) and 1(d) illustrate the high-resolution TEM images of the three stacked grown InAs/GaAs QD layers and a single QD, respectively. Figure 1(e) shows the atomic distribution profile of In as a function of depth from the surface on the epitaxial chip obtained by x-ray photoelectron spectroscopy (XPS). The QDs monolithically grown on Si present good uniformity with a density of  $\sim 4 \times 10^{10} \text{ cm}^{-2}$ , of which the typical size is 25 nm in  $D$  and 8 nm in height. Moreover, the threading dislocation density was estimated at around  $10^7$  to  $10^8 \text{ cm}^{-2}$ , determined by atomic force microscopy (AFM) [Fig. 1(f)] and TEM images. Figure 1(g) demonstrates the room-temperature photoluminescence (PL) spectra of the as-grown structure at a range of input power from 3.5  $\mu\text{W}$  to 220  $\mu\text{W}$ , indicating ground state emission (peak at around 1315 nm) was at 1.3  $\mu\text{m}$  telecommunication wavelength band. PL from the excited states and the wetting layer of QDs was observed when increasing the pump power.

Various microdisk patterns were defined by using electron-beam lithography with a ZEP520 electron beam resist. A layer of silicon dioxide (SiO<sub>2</sub>) with a thickness of around 120 nm was first deposited on the wafer by plasma-enhanced chemical vapor deposition and used as the hard mask. The microdisk patterns were transferred from the resist into the hard mask using reactive ion etching (RIE). After removing the resist, the hard mask patterns were further transferred through the active region using inductively coupled plasma RIE (ICP-RIE). Then wet etching was used to remove the SiO<sub>2</sub> hard mask and form the supporting pedestal. The fabricated microdisk lasers were characterized in a micro-PL ( $\mu\text{-PL}$ ) system at room temperature and were CW optically pumped using a 632.8 nm He-Ne laser with focus spot size of  $\sim 3 \mu\text{m}$ .



**Fig. 1.** (a) TEM image of the QD structure grown on on-axis Si (001) substrate. (b) Schematic illustration of the epitaxial structure of the active region. (c), (d) High-resolution TEM images of the three stacked InAs QD layers and a single QD, respectively. (e) Atomic distribution profile of In obtained by XPS. (f) AFM image of uncapped InAs/GaAs QDs grown on Si (001) substrate. (g) Room-temperature PL spectra of the as-grown structure at various input powers.

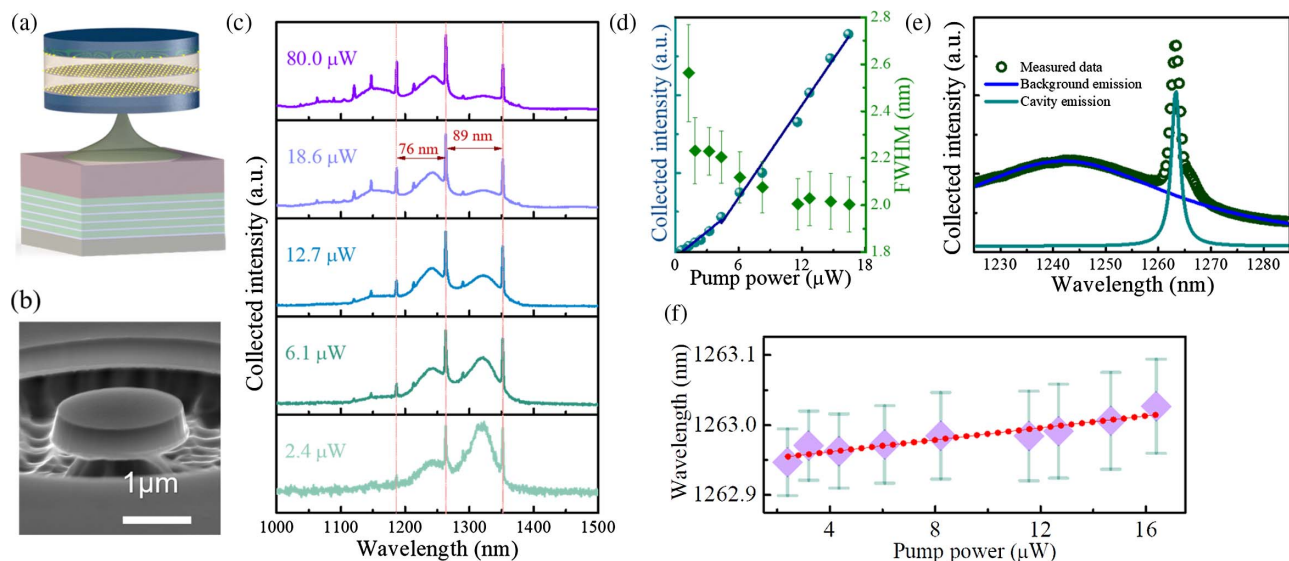
### 3. RESULTS AND DISCUSSION

Figure 2(a) shows a schematic diagram of the fabricated microdisk cavity, with three layers of QD embedded inside the active region. A scanning electron microscope (SEM) image of the fabricated microdisk cavity with  $D \sim 1.9 \mu\text{m}$  is shown in Fig. 2(b), indicating a smooth surface of the active region and  $\sim 73.5^\circ$  sidewall tilt. Figure 2(c) shows the collected PL spectra below and above the lasing threshold of the microdisk with  $D \sim 1.9 \mu\text{m}$ , indicating a measured free spectral range (FSR) of  $\sim 76 \text{ nm} - 89 \text{ nm}$  between the adjacent WGMs in the same radial order. The measured FSR is consistent with the calculated value (calculated value:  $\text{FSR} = \frac{\lambda^2}{\pi D n_{\text{eff}}} \sim \frac{(1.250 \mu\text{m})^2}{\pi (1.9 \mu\text{m})(3.4)} = 77 \text{ nm}$ ). Lasing emissions from both the ground state and excited states were obtained, from which the main peak (1263 nm) was located within the first excited state. Excited states lasing emission was also observed for microdisk lasers grown on V-grooved-Si substrate with five stacked InAs QDs [20,21]. The corresponding collected intensity (L-L curve) and full width at half maximum (FWHM) for the emission peak at 1263 nm as a function of input power are demonstrated in Fig. 2(d), which shows the ultra-low threshold of  $2.6 \pm 0.4 \mu\text{W}$  obtained. (Note that due to the multimode lasing behavior, the lasing threshold of the emission peak at 1263 nm should be lower than  $2.6 \pm 0.4 \mu\text{W}$ ; this threshold error came from both the measured power and the curve fitting of measured PL data.) The lasing threshold was determined by the L-L curve. The trend of the FWHM narrowing was obvious, and this was used to judge the lasing behavior. The measured lasing threshold was even lower than those of the InAs QD microdisk lasers directly grown on GaAs or InP substrates [19,42,43]. During the  $\mu\text{-PL}$  measurements, both the broad background and resonant emissions were observed. Background emission can be suppressed by enhancing the coupling between the emitter and microcavity and improving the spatial overlap between the gain material and WGMs. Figure 2(e) demonstrates the curve fitting of the measured spectra above the threshold ( $4.4 \mu\text{W}$ ), which comprises

a spontaneous emission redshift background and cavity emission. Figure 2(f) displays the redshift of the measured lasing peaks with increasing incident pump power induced by thermal effects. A redshift rate of  $d\lambda/dP_{\text{pump}} \sim 5.69 \text{ nm/mW}$  was obtained using a linear fit.

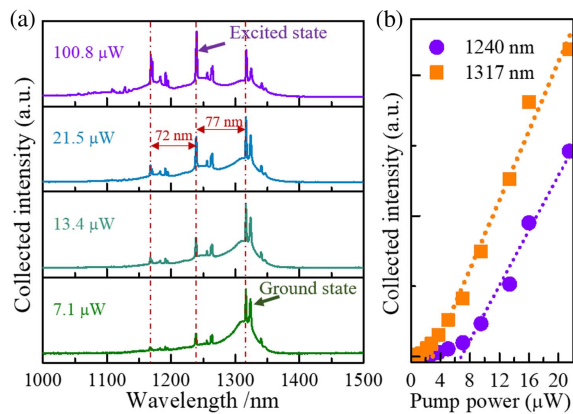
By increasing the pump power, the intensity of excited states emission can be stronger than that of the ground state emission, as shown in Fig. 1(g). The higher pump power reveals mode competition between the ground state and excited states of QDs due to gain saturation. Figure 3(a) shows the collected laser spectra of a microdisk laser with  $D \sim 2 \mu\text{m}$  under various pump powers. Increasing the pump power results in gradual gain saturation of the ground state transition (1317 nm), and lasing switches to the excited state (1240 nm). The ground state gain saturation can be further prevented by increasing the stacked layers of QDs within the active layer [44]. The thresholds of lasing peaks at 1317 nm and 1240 nm are  $2.7 \pm 0.4 \mu\text{W}$  and  $6.5 \pm 0.4 \mu\text{W}$ , respectively, indicated by the L-L curve [Fig. 3(b)]. The relatively higher lasing threshold of the excited state is a consequence of the larger density of excited states.

One way to obtain single-mode lasing emission is to decrease the  $D$  of the microdisk cavity. The FSR becomes larger with a smaller  $D$  of the microdisk cavity, which leads to well-separated resonant peaks for single-mode lasing emission (i.e., the FSR is comparable to or larger than the FWHM of the gain spectra). Here, the lasing characteristics of a microdisk laser with  $D \sim 1.4 \mu\text{m}$  and a sub-wavelength microdisk laser with  $D \sim 1.1 \mu\text{m}$  were measured. Figure 4(a) presents the lasing spectra of the microdisk lasers with  $D \sim 1.1 \mu\text{m}$  and  $\sim 1.4 \mu\text{m}$ , in which both the first- and second-order WGMs are observed with increasing pump power. A much broader measured FSR of 159 nm for the microdisk laser with  $D \sim 1.1 \mu\text{m}$  (FSR of 116 nm for the  $D \sim 1.4 \mu\text{m}$ ) of the same radial order is observed compared with the FSR presented in Figs. 2(c) and 3(a). A side-mode suppression ratio of 6.5 dB is obtained. The FSR is also much larger than the



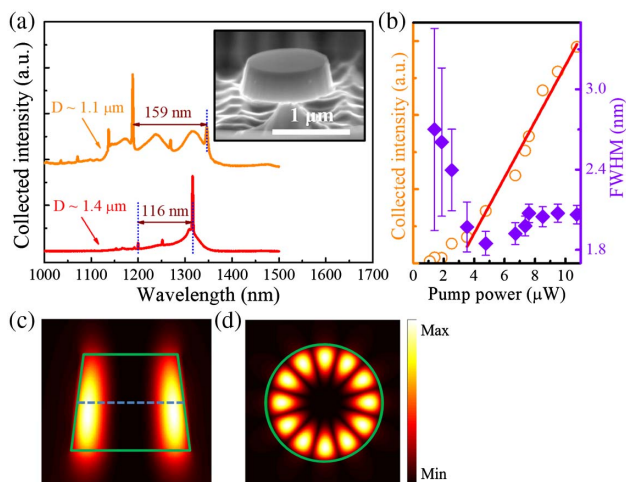
**Fig. 2.** (a) Schematic diagram of a microdisk laser grown on planar on-axis Si (001) substrate. (b) SEM image of a fabricated microdisk laser. (c) Collected PL spectra below and above the lasing threshold of a microdisk with  $D \sim 1.9 \mu\text{m}$ . (d) Corresponding collected intensity and mode linewidth as a function of input power for the emission peak at 1263 nm. The lasing threshold is  $\sim 2.6 \pm 0.4 \mu\text{W}$ . (e) Collected spectra above the threshold ( $4.4 \mu\text{W}$ ) and curve fitting showing the spontaneous emission background and cavity emission. (f) Measured lasing wavelength under various incident pump powers.



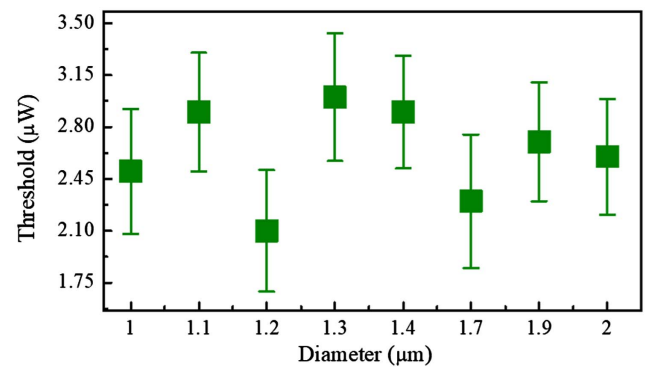


**Fig. 3.** (a) Measured lasing spectra of the microdisk laser with  $D \sim 2 \mu\text{m}$  under various pump powers. Lasing emissions from both ground state and excited states are observed. (b) L-L curve of the emission peaks at 1240 nm (first excited state) and 1317 nm (ground state), respectively.

linewidth ( $\sim 46 \text{ nm}$ ) of the ground state emission, which substantially can support one resonate frequency of first-order WGMs within the ground state. In addition, the mode positions (cavity resonate frequencies) are highly dependent on the structural parameters of the cavity. During the optical pumping process, the lasing emission peak from the excited states at 1189 nm (ground state at 1315 nm) dominates the lasing spectra of the microdisk laser with  $D \sim 1.1 \mu\text{m}$  ( $D \sim 1.4 \mu\text{m}$ ), as shown in the Fig. 4(a). A lasing peak with weaker intensity of the ground state is also observed for the microdisk laser with  $D \sim 1.1 \mu\text{m}$ , for which there is not clear mode switching from the ground state to the excited states with the gradual increase in pump power. The lasing in the excited states is due mainly to the mode selection of the sub-wavelength scale microdisk cavity. The inset in Fig. 4(a)



**Fig. 4.** (a) Measured lasing spectra of the microdisk laser with  $D \sim 1.1 \mu\text{m}$  and  $\sim 1.4 \mu\text{m}$ . The inset SEM image shows the fabricated sub-wavelength scale microdisk laser. (b) L-L curve and FWHM of lasing peak  $\sim 1189 \text{ nm}$  of the sub-wavelength scale microdisk laser with  $D \sim 1.1 \mu\text{m}$ , showing the lasing threshold of  $\sim 2.9 \pm 0.4 \mu\text{W}$ . (c), (d) Calculated cross section and top view of the magnetic field profiles for the  $\text{TE}_{1,6}$  mode, respectively. The green line represents the boundary of the microdisk, and the blue dashed line demonstrates the central plane of the microdisk.



**Fig. 5.** (a) Thresholds of microdisk lasers with various  $D$ . The lasing threshold was derived from the L-L curve of the main lasing peak, and all measured data came from a single microdisk laser.

shows the SEM image of a fabricated sub-wavelength scale microdisk laser. The L-L curve and FWHM of the lasing peak at 1189 nm for the microdisk laser with  $D \sim 1.1 \mu\text{m}$  are shown in Fig. 4(b), indicating the lasing threshold of  $\sim 2.9 \pm 0.4 \mu\text{W}$ . The mode profile of lasing peak at 1189 nm was calculated using the 3D finite-difference time-domain (3D-FDTD) method, and was identified as  $\text{TE}_{1,6}$ . Figures 4(c) and 4(d) show, respectively, the cross section and top view of the calculated magnetic field profiles for  $\text{TE}_{1,6}$ . The green line indicates the boundary of the microdisk resonator, and the blue dashed line demonstrates the central plane of the microdisk. As indicated in Fig. 4(c), the central plane of WGMs is not completely overlapped with the emitter, which may cause the broad background emission of the lasing spectra.

In addition, the thresholds of microdisk lasers with various  $D$  from  $1 \mu\text{m}$  to  $2 \mu\text{m}$  are presented in Fig. 5, which indicates low thresholds below  $3.5 \mu\text{W}$ . The lasing threshold of microdisk lasers with smaller  $D$  was assumed to be lower due to the decreased gain region. However, the measured lasing thresholds fluctuated with the  $D$ , which was highly dependent on the fabrication process factors, such as the width of the supporting pedestal of the microdisk and surface roughness. In addition, the thresholds could have been influenced by the non-uniform distribution of as-grown QDs as well as the difference in gain coefficients of the ground state and excited states.

#### 4. CONCLUSION

In conclusion, we developed InAs/GaAs QD microdisk lasers monolithically grown on on-axis Si (001) substrate. The microdisk lasers were CW optically pumped at room temperature, and an ultra-low lasing threshold of  $\sim 3 \mu\text{W}$  with  $D \sim 1.1 \mu\text{m}$  was obtained. Lasing emissions from both the ground state and excited states were demonstrated. The promising lasing characteristics of the microdisk lasers monolithically grown on Si (001) substrate with an ultra-low threshold and small footprint provide a viable route towards the large-scale, low-cost integration of laser sources on a silicon platform.

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