

# **Factors Governing the Enhancement of Hydrocarbon Recovery via H<sub>2</sub>S and/or CO<sub>2</sub> Injection: Insights from a Molecular Dynamics Study in Dry Nano-Pores**

**Sakiru B. Badmos, Tai Bui, and Alberto Striolo\***

Department of Chemical Engineering  
University College London  
London, WC1E 7JE, UK

**David R. Cole**

School of Earth Sciences  
The Ohio State University  
Columbus, Ohio 43210, USA

## **ABSTRACT**

Although enhanced oil recovery (EOR) is often achieved by CO<sub>2</sub> injection, the use of acid gases has also been attempted, for example in oil fields in west Canada. To design EOR technologies effectively, it would be beneficial to quantify the molecular mechanisms responsible for enhanced recovery under various conditions. We report here molecular dynamics simulation results that probe the potential of recovering *n*-butane confined from silica, muscovite and magnesium oxide nano-pores, all proxies for subsurface materials. The three model solid substrates allow us to identify different molecular mechanisms that control confined fluid behavior, and to identify the conditions at which different acid gas formulations are promising. The acid gases considered are CO<sub>2</sub>, H<sub>2</sub>S, as well as their mixtures. For comparison, in some cases we consider the presence of inert gases such as N<sub>2</sub>. In all cases, the nano-pores are dry. The recovery is quantified in terms of the amount of *n*-butane displaced from the pore surface as a function of amount of gases present in the pores. The results show that the gas performance depends on the chemistry of the confining substrate. While CO<sub>2</sub> is more effective at displacing *n*-butane from the protonated silica pore surface, H<sub>2</sub>S is more effective in muscovite, and both gases show similar performance in MgO. Analysis of the interaction energies between the confined fluid molecules and the surface demonstrates that the performance depends on the gas interaction with the surface, which suggests experimental approaches that could be used to formulate the gas mixtures for EOR applications. The structure of the gas films at contact with the solid substrates is also quantified, as well as the self-diffusion coefficient of the fluid species in confinement. The results could contribute to designing strategies for achieving both improved hydrocarbon production and acid gas sequestration.

**Keywords:** Preferential adsorption, Structure-transport relation, Adsorption

\* Corresponding Author: a.striolo@ucl.ac.uk

## 1.0 INTRODUCTION

The growing concern about greenhouse gas emissions has stimulated research into developing new technologies for capturing and sequestering CO<sub>2</sub>. CO<sub>2</sub> injection into geological formations has received much attention,<sup>1-8</sup> sometimes as a long-term storage opportunity, while in some other cases CO<sub>2</sub> has been injected in oil and gas fields to attempt to simultaneously achieve enhanced hydrocarbon recovery and CO<sub>2</sub> sequestration.<sup>9</sup> For these strategies to be fully optimised, it is important to understand, and ultimately control the molecular mechanisms that are responsible for rock-fluid interactions, inclusive of fluid sorption, migration, and fixation.<sup>10</sup>

We consider here enhanced oil recovery (EOR). EOR methods commonly used include water, gas, and surfactant injection.<sup>11-14</sup> The technology follows either the flooding or the huff-n-puff mode. The huff-n-puff mode is sometimes preferred for shale formations due to their ultra-low permeability, which delays gas and pressure propagation from the injector to the producer well.<sup>15</sup> It has been reported that waterflooding can yield much lower recovery compared to gas injection, which sometimes justifies the use of gas injection as the preferred option.<sup>16</sup>

Several studies have been reported in the literature concerning efforts directed towards understanding the mechanisms responsible for CO<sub>2</sub>-based EOR.<sup>7-8, 14, 17-23</sup> In general, the enhancement of hydrocarbon recovery is either attributed to preferential adsorption of CO<sub>2</sub> on the pore surfaces,<sup>8, 24-26</sup> or the dissolution of CO<sub>2</sub> in oil, which swells the oil and reduces its viscosity.<sup>6, 27-29</sup> Previous research from our group<sup>24</sup> reported that CO<sub>2</sub> preferentially adsorbs on silica surfaces, weakening *n*-butane adsorption, and effectively acting as a ‘molecular lubricant’ that lowers the activation energy for *n*-butane diffusion. Santos et al.<sup>21</sup> conducted molecular dynamics simulations in slit-shaped calcite pores to study the effect of CO<sub>2</sub> on *n*-alkanes displacement. They identified several factors that control the preferential adsorption of CO<sub>2</sub> on calcite, including the amount of CO<sub>2</sub> present, temperature, hydrocarbon length, and pore size. Wang et al.<sup>22</sup> observed an increase in *n*-decane diffusion as CO<sub>2</sub> loading increases, and reported a maximum diffusion due to pore crowding.

Experimental studies on CO<sub>2</sub> – based EOR are also common.<sup>30-33</sup> Jin et al.<sup>30</sup> carried out experimental studies on core samples from the Bakken shale formation, and found that supercritical CO<sub>2</sub> injection

facilitates the recovery of up to 65% of hydrocarbons in place; they also reported that CO<sub>2</sub> is trapped in the reservoir over a wide pressure range. Eide et al.<sup>31</sup> reported experimental results on oil recovery by CO<sub>2</sub> injection into fractured core sample using nuclear magnetic resonance (NMR) and X-ray computed tomography; they reported oil recovery in excess of 90% of the original oil in place.

Current research focuses on coupling EOR with CO<sub>2</sub> sequestration. However, sequestering H<sub>2</sub>S resulting from oil and gas processing into geological formations is also desirable, and could lead to EOR. Acid gases (CO<sub>2</sub>+H<sub>2</sub>S) have been injected into geological formations to reduce atmospheric emissions and at the same time enhance hydrocarbon recovery.<sup>34</sup> Field tests have been reported where acid gases were injected into geological formations in the Alberta basin of western Canada.<sup>35</sup> However, Khan et al.<sup>34</sup> performed reservoir simulation studies complemented by experimental data from Clean Gas Technology Australia (CGTA) as input; based on those reservoir simulations, injecting a mixture of CO<sub>2</sub> and H<sub>2</sub>S yields a lower recovery factor compared to injecting pure CO<sub>2</sub>.

Because it is possible that the performance of an EOR strategy depends on the formation features (rocks, pressure, temperature, hydrocarbons in place, porosity, presence of other fluids such as water, etc.), a detailed molecular-level understanding of the mechanisms responsible for EOR is desirable. This could be achieved by extensive molecular simulations. While results have been reported for CO<sub>2</sub>, CH<sub>4</sub> and other hydrocarbons in nanopores, few studies explicitly attempted to quantify the behaviour of H<sub>2</sub>S and its mixtures with CO<sub>2</sub> within narrow pores.<sup>36</sup> An exception is our recent study,<sup>37</sup> where we reported that H<sub>2</sub>S solubility in water confined in narrow pores is much lower compared to the H<sub>2</sub>S solubility in bulk water because confinement in nano-pores can strongly affect the hydration shell of aqueous H<sub>2</sub>S.

In this manuscript, we investigate the *n*-butane displacement due to pure H<sub>2</sub>S, pure CO<sub>2</sub> as well as their mixtures within slit-shaped nano-pores carved out of silica, muscovite, and magnesium oxide (MgO). The three pores are chosen because silica and muscovite are considered representative of many subsurface formations, while MgO is a model substrate useful for quantifying the molecular phenomena responsible for the results obtained. It should be noted that the MgO surface considered here is not hydroxylated. As such, our system provides a model surface useful for understanding the molecular driving forces at play for the systems considered, but it does not provide a good model for real

substrates, e.g. brucite. Because the systems considered here are dry, pH effects on the pore surfaces are not described. An exception is the silica substrate, which is protonated in our model. Certainly, the presence of water would affect the results presented, and future studies, conducted perhaps with reactive force fields, should address such effects. We focus on the molecular behaviour of the fluids at the solid-fluid interface, as well as the transport of the confined fluids. We employ the method of atomistic molecular dynamics (MD) simulations to probe the effect of molecular interactions on the results obtained. While EOR performance cannot be directly quantified by the MD approach implemented here, the results are analysed in terms of the preferential adsorption of *n*-butane vs. acid gases on the solid surfaces. The implicit assumption is that the dislocation of *n*-butane from the proximity to the solid substrates promotes EOR.

The remainder of this manuscript is organized as follows: in Section 2 we describe the simulation models and algorithms implemented in this study; in Section 3 we present the simulation results; we then conclude by briefly discussing and summarizing our main findings.

## **2 SIMULATION MODELS AND METHODOLOGY**

### **2.1 Simulation set up**

Molecular dynamics (MD) simulations were conducted for binary and ternary fluid systems composed of systems of C<sub>4</sub>H<sub>10</sub>-H<sub>2</sub>S, C<sub>4</sub>H<sub>10</sub>-CO<sub>2</sub>, C<sub>4</sub>H<sub>10</sub>-N<sub>2</sub> and C<sub>4</sub>H<sub>10</sub>-H<sub>2</sub>S-CO<sub>2</sub> at various compositions confined within slit-shaped pores of width 22Å carved out of silica, muscovite, and MgO. These substrates are representative of many minerals or mineral components found in sub-surface formations. The silica surface was obtained by cutting  $\beta$ -cristobalite crystal along the (1,1,1) crystallographic plane; the non-bridging oxygen atoms were fully protonated yielding a reasonable proxy for hydrophilic surfaces.<sup>38</sup> The resulting -OH surface density is 4.54 per square nanometer. The model MgO slab was obtained from the space group Fm3m replicated along the (001) plane.<sup>39-40</sup> The exposed MgO surface was not hydroxylated in our model. While this model is not realistic, it allows us to quantify the mechanisms responsible for preferential adsorption. Muscovite was considered a model for clays, used in our prior investigations. Details regarding this substrate are available elsewhere.<sup>41-43</sup> Muscovite is a phyllosilicate

mineral with structure similar to illite.<sup>44</sup> In muscovite, an interlayer of potassium ions holds a Tetrahedral-Octahedral-Tetrahedral (T-O-T) structure of Al-centred octahedral sheets sandwiched between two Si-centred tetrahedral sheets in which one Al atom substitutes one out of every four Si atoms. The potassium ion-bearing interlayer balances the negative charge and holds the T-O-T layers via electrostatic interactions.<sup>41</sup> Each octahedral sheet contains two oppositely pointing –OH groups. The surface of each muscovite slab is not protonated in our studies, but it contains potassium ions resulting from cleavage along its basal plane (001). Muscovite with this surface termination has been used in several studies.<sup>41-43, 45-47</sup> No water is considered in the present study, although it would certainly affect the results, for example via the solvation of the K<sup>+</sup> ions.

All solid substrates simulated bear no net charge and were kept rigid throughout the simulation. Exceptions were the surface hydrogen atoms in silica and potassium ions on muscovite, which were allowed to vibrate. Each solid slab was maintained parallel to the X-Y plane of the simulation box. The X and Y dimensions for the three systems were 51.7 x 100.8, 47.2 x 73.4, and 47.2 x 73.6 Å<sup>2</sup> for silica, muscovite and MgO, respectively. The Z dimension of the simulation box was set to 54.92, 58.30 and 53.52 Å for silica, muscovite, and MgO, respectively. These dimensions yield slit-shaped pores 22Å wide, measured as the shortest centre-to-centre distance between the oxygen atoms of the –OH groups on silica slabs, surface potassium ions in muscovite slabs and surface magnesium atoms in MgO across the pore volume. Because periodic boundary conditions (PBC) were applied in all directions, the simulated systems are infinitely long in X and Y directions.

The pores were loaded with 300 molecules of n-butane. It is estimated that this amount of hydrocarbon corresponds to approximately 60% of the maximum loading in silica pores at 350K. We report in **Supporting Information** details regarding these simulations, and the maximum amount of n-butane, CO<sub>2</sub> and H<sub>2</sub>S estimated to be confined within the pores used in his work at moderate pressures.

Binary systems containing n-butane and H<sub>2</sub>S were simulated at different compositions: 300 molecules of n-butane were diluted with H<sub>2</sub>S at different loadings. These are systems 1-3 in **Table 1**. Note that the number of n-butane molecules is maintained constant for all these simulations. Snapshots for n-butane-H<sub>2</sub>S systems for the maximum H<sub>2</sub>S loading are shown in **Figure 1**.

Binary systems containing *n*-butane-CO<sub>2</sub> and *n*-butane-N<sub>2</sub> were also simulated at different compositions: 300 molecules of *n*-butane were diluted with CO<sub>2</sub> and with N<sub>2</sub> at different loadings. These are systems 4-6 and 7-9, respectively, in **Table 1**. Representative snapshots for *n*-butane-CO<sub>2</sub> and *n*-butane-N<sub>2</sub> systems confined in the three pores are shown in **Supporting Information (SI)**.

To study the effect of H<sub>2</sub>S-CO<sub>2</sub> mixtures and the presence of an inert gas (nitrogen) on the displacement of *n*-butane from the pore surface, we simulated ternary systems of *n*-butane-H<sub>2</sub>S-CO<sub>2</sub> with varying compositions of the acid gases. These are systems 10-12 in **Table 1**. *n*-butane-acid gas-nitrogen (N<sub>2</sub>) systems are identified as 13-16 in **Table 1**.

As the system composition and the amount of fluid molecules confined in the pores change, so does the pressure. To relate the system composition to bulk pressure, we performed simulations in which the pores are in contact with bulk reservoirs. Constraining the number of *n*-butane in the pores, CO<sub>2</sub> and H<sub>2</sub>S molecules were allowed to exchange between the pores and the reservoirs. The bulk pressure was calculated from the density in the reservoirs, as explained in the Supporting Information (Figure S1). In **Table 2** we report the pressure ranges corresponding to the systems summarised in **Table 1**. Pressures for muscovite and MgO pores were similar, while silica pores were at somewhat lower pressures.

**Table 1:** Composition of all systems simulated in this work.

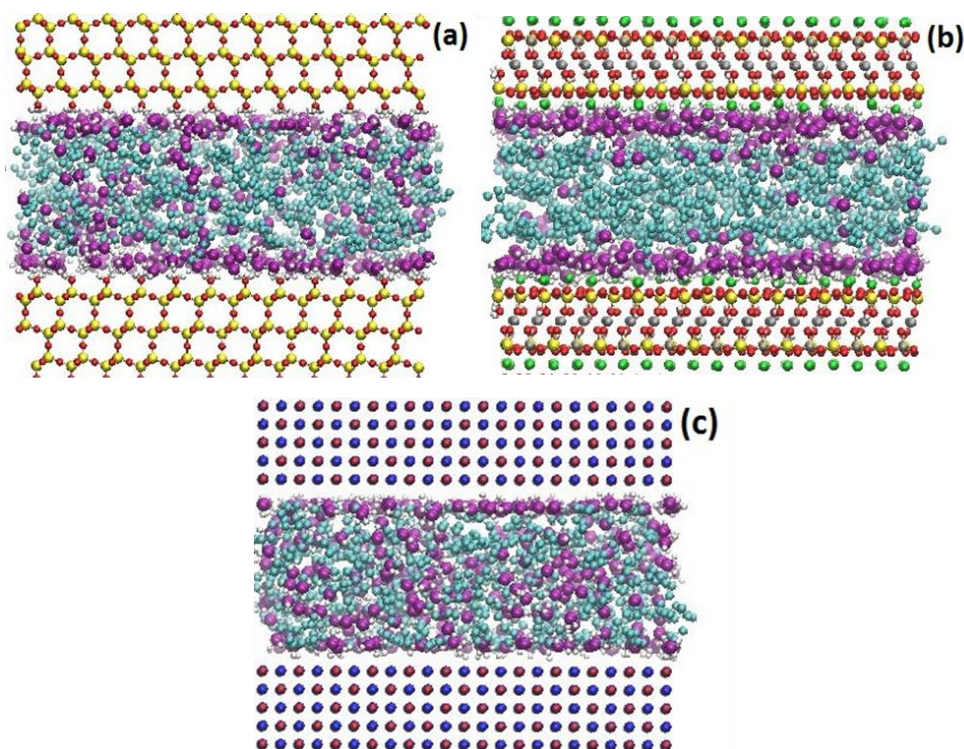
System	Number of <i>n</i> -butane	Number of H <sub>2</sub> S	Number of CO <sub>2</sub>	Number of N <sub>2</sub>
1	300	200	-	-
2		375	-	-
3		500	-	-
4		-	200	-
5		-	375	-
6		-	500	-
7		-	-	300
8		-	-	500
9		-	-	700
10		250	250	-
11		125	250	-
12		250	125	-
13		400	-	100
14		300	-	200
15			400	100
16			300	200

**Table 2:** Bulk Pressures corresponding to the systems summarised in Table 1. For each pore, we report the lower and upper pressures, as estimated at increasing acid gas loading. For details, please refer to Figure S1 of the SI.

<b>Substrate</b>	<b>Pressure / MPa (n-butane with_H<sub>2</sub>S)</b>	<b>Pressure / MPa (n-butane with_CO<sub>2</sub>)</b>
Silica	14.2 ± 0.2 – 7.8 ± 0.3	14.0 ± 0.2 – 7.6 ± 0.3
Muscovite	21.0 ± 0.2 – 11.2 ± 0.2	19.2 ± 0.1 – 10.9 ± 0.2
MgO	23.2 ± 0.2 – 12.1 ± 0.3	20.1 ± 0.1 – 11.2 ± 0.2

## 2.2 Force fields

The CLAYFF force field was implemented to model silica, muscovite, and MgO.<sup>48</sup> *n*-Butane, and CO<sub>2</sub> were modeled using the TraPPE-UA force field.<sup>49</sup> In TraPPE-UA, CO<sub>2</sub> is rigid with all atoms on a straight line forming a bond angle of 180°. *n*-Butane is flexible, described by bond stretching, angle bending and dihedrals. The united-atom formalism was implemented to describe -CH<sub>3</sub> and -CH<sub>2</sub> groups of *n*-butane. H<sub>2</sub>S was described by the model developed by Kamath and Potoff.<sup>50</sup> Nitrogen (N<sub>2</sub>) was described as a single LJ sphere without Coulombic interactions.<sup>51</sup> The N<sub>2</sub> model implemented here was found to reasonably reproduce the experimental adsorption isotherm of N<sub>2</sub> in silica pores.<sup>51</sup> In all cases, non-bonded interactions were modeled by dispersive and electrostatic interactions. The electrostatic interactions were modeled by Coulombic potential and the dispersive interactions were described by 12-6 Lennard-Jones (LJ) potentials. The LJ parameters for unlike atoms were obtained using Lorentz-Berthelot combination rules.<sup>52</sup> The cut-off distance for all interactions was set to 14Å. The particle mesh Ewald method was implemented for long-range corrections to electrostatic interactions.<sup>53</sup>



**Figure 1:** Simulation snapshots for binary systems containing *n*-butane and H<sub>2</sub>S in silica (a) MgO (b) and muscovite (c). All systems shown contain 300 *n*-butane and 500 H<sub>2</sub>S molecules. Cyan spheres are -CH<sub>3</sub> and -CH<sub>2</sub> in *n*-butane, purple are sulphur, white are hydrogen, red are oxygen, yellow are silicon, green are potassium, grey are aluminium, blue are magnesium. For clarity, only a portion of the solid substrates are shown along the Z direction. Please refer to Section 2.1 for details regarding the size of the simulation boxes.

### 2.3 Algorithms

All simulations were performed within the canonical ensemble, in which the number of molecules, volume, and temperature were maintained constant (NVT). The simulation package GROMACS, version 5.1.2,<sup>54-55</sup> was used for conducting the simulations. Numerical integration of Newton's equations of motion was performed using the leapfrog algorithm with a time step of 1fs. The temperature of the solid substrates and of the fluid molecules were controlled separately using two Nosé-Hoover thermostats with relaxation times of 200fs. The simulations were conducted at 350K, a temperature representative of depths between 11,000 and 15,000 feet in sub-surface formations. The total simulation time for each system was in the range of 60-80 ns, depending on the system and the loading. The system was considered equilibrated when *n*-butane and acid gas densities fluctuated around a constant value, and the system energy fluctuated within 10% of the average values. In Figure



S2 of the SI we report representative data for the interaction energy as a function of simulation time for the systems considered here. The results confirm that equilibration was achieved. The production phase was conducted at the end of the simulations, and data analysis was conducted over the last 10 ns of each simulation.

### 3 RESULTS AND DISCUSSION

#### 3.1 Binary Systems: Enhancement of Butane Displacement by H<sub>2</sub>S and CO<sub>2</sub> in Silica Pores

##### 3.1.1 Density profiles

The distribution of fluid molecules in the pore is quantified in terms of the molecular density profiles along the direction perpendicular to the pore surface. The density profiles calculated for the centre of mass (COM) of *n*-butane at different H<sub>2</sub>S, CO<sub>2</sub> and N<sub>2</sub> loadings are shown in **Figure 2**. The density profile results show preferential adsorption of *n*-butane on the pore surfaces, as evidenced by the high peak density. The height of the *n*-butane density peak closest to the solid substrate reduces as the H<sub>2</sub>S density is increased, suggesting that the amount of *n*-butane in contact with the pore surface decreases. The results show that adding CO<sub>2</sub> also reduces the intensity of the density peak closest to the solid substrate. Conversely, adding N<sub>2</sub> leaves the density profile relatively unchanged, as discussed shortly.

The density profiles of the hydrogen and sulphur atoms of H<sub>2</sub>S, of the carbon and oxygen atoms of CO<sub>2</sub>, and of N atom of N<sub>2</sub> in the binary systems considered in **Figure 2** are shown in **Figure 3**. The distribution of H<sub>2</sub>S in the pore is similar to that of CO<sub>2</sub> in the binary systems. At low loading, most of the H<sub>2</sub>S and CO<sub>2</sub> in the pore are adsorbed close to the pore surface while at higher loadings, some H<sub>2</sub>S and CO<sub>2</sub> molecules occupy regions close to the pore centre, as the pore surface becomes more saturated.

To quantify these results, we calculated the percentage of *n*-butane displaced from the first adsorbed layer (FAL) by:

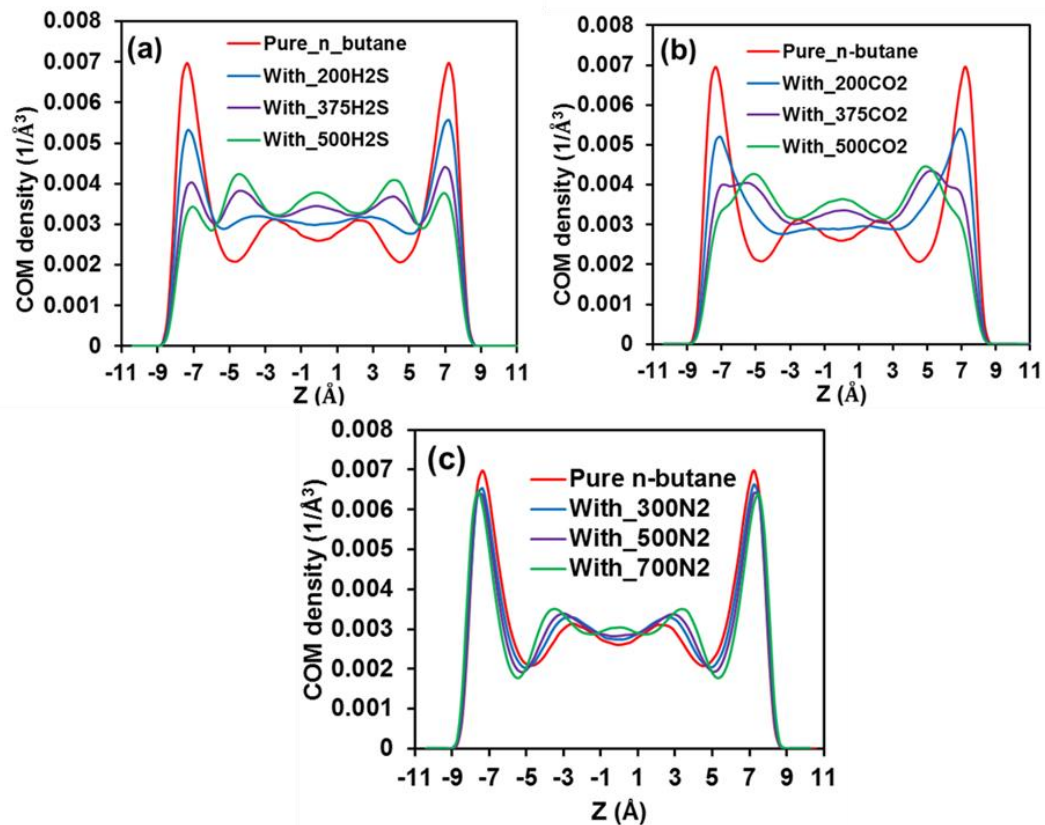
$$\eta_{displaced} = \frac{(N_{initial} - N_{gas})}{N_{initial}} \quad (1)$$

In Eq. (1),  $N_{initial}$  and  $N_{gas}$  represent the amount of *n*-butane in the FAL in the system without acid gas and with acid gas respectively.  $\eta_{displaced}$  is calculated within the narrow region confined between the

position of the first peak in *n*-butane COM density profile and the pore surface. The results, obtained from the density profiles, are shown in **Table 3** and confirm that both CO<sub>2</sub> and H<sub>2</sub>S aid displacing *n*-butane from the pore surfaces. It appears that CO<sub>2</sub> is slightly more effective than H<sub>2</sub>S.

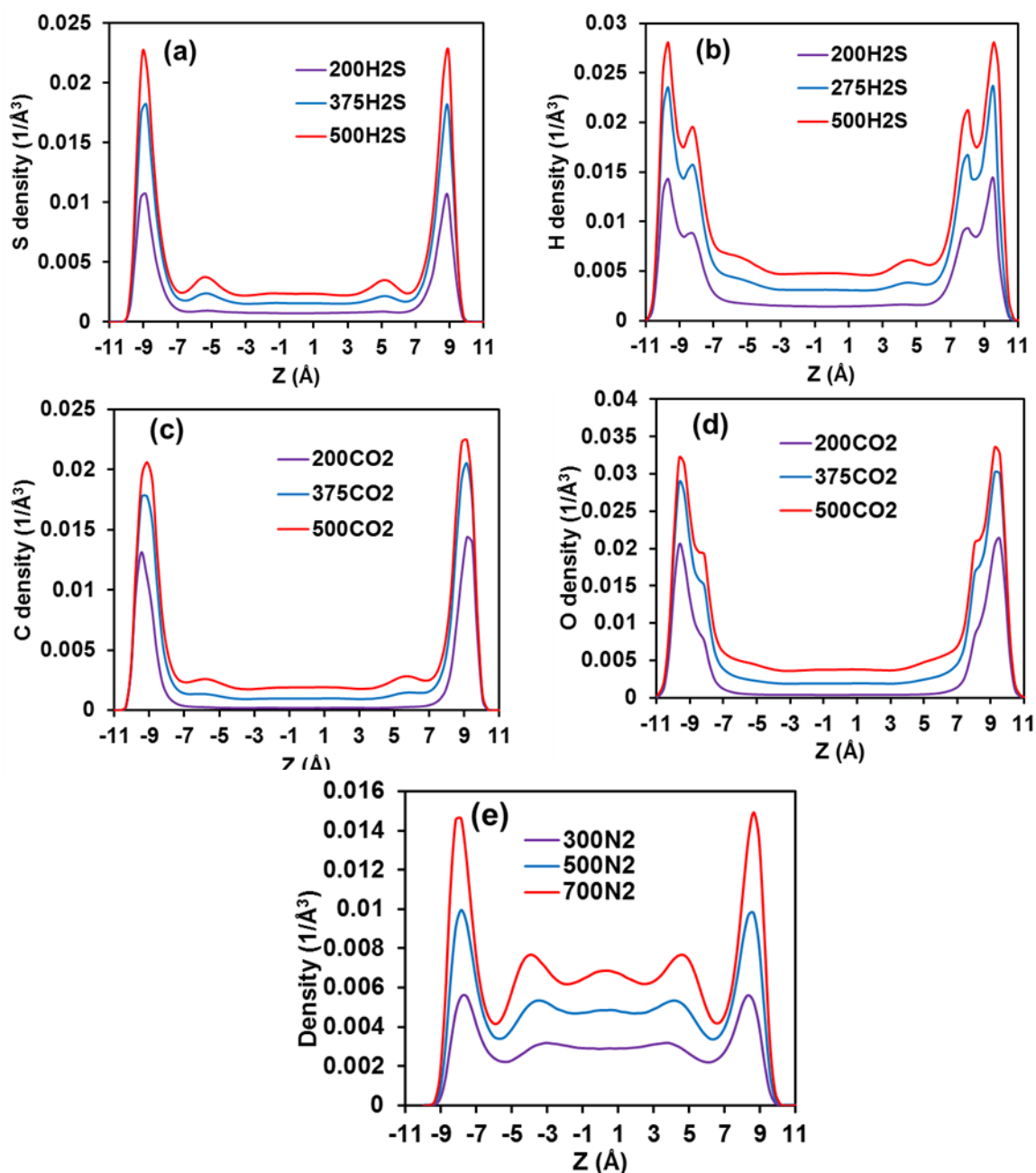
**Table 3:** Percentage of *n*-butane displaced with H<sub>2</sub>S and CO<sub>2</sub> loading. Uncertainties, estimated by block averaging the simulation results, are  $\sim 1$  in all cases.

Loading	% displaced (H <sub>2</sub> S loading)	% displaced (CO <sub>2</sub> loading)
200	28	34
375	48	55
500	58	65



**Figure 2:** Density profile of the COM of *n*-butane in the binary system with H<sub>2</sub>S loading (a), CO<sub>2</sub> loading (b) and N<sub>2</sub> loading (c). The density profile for a system with pure *n*-butane is also shown for comparison. All systems contain 300 *n*-butane molecules.

For completeness, we also simulated systems of *n*-butane in silica pores in the presence of N<sub>2</sub>, without acid gases. The density profiles of COM of *n*-butane and N of N<sub>2</sub> for the binary systems are shown in **Figure 2c** and **3e**, respectively. The *n*-butane density profiles remain almost unchanged despite the presence of N<sub>2</sub>. Comparing the density profiles of CO<sub>2</sub> and H<sub>2</sub>S in **Figure 3** to that of N<sub>2</sub> in **Figure 3e** shows that the acid gases are more strongly adsorbed on the silica surface than N<sub>2</sub> as the peak heights within the FAL are higher. The density profile of N<sub>2</sub> in **Figure 3e** shows that more N<sub>2</sub> occupy the middle of the pore, where it probably mixes with *n*-butane and potentially reduces its viscosity. To test this latter possibility, we simulated bulk systems with densities approximately equal to those in the middle of the pore. We calculated the viscosity using the procedure described elsewhere,<sup>56</sup> and we obtained 0.16cP for pure *n*-butane and 0.21 and 0.24cP for *n*-butane-N<sub>2</sub> systems containing 200 *n*-butane and 200 N<sub>2</sub> molecules. These results suggest that N<sub>2</sub> increases the viscosity in the simulated pores, which is opposite of what was expected. This however is probably a consequence of the fact that the confined system becomes denser upon N<sub>2</sub> addition. Infact, the viscosity increases with pressure, as shown in **SI**.



**Figure 3:** Density profile of S of  $\text{H}_2\text{S}$  (a) H of  $\text{H}_2\text{S}$  (b) C of  $\text{CO}_2$  (c) O of  $\text{CO}_2$  (d) and N of  $\text{N}_2$  (e) in the binary systems at different acid gas and  $\text{N}_2$  loadings in silica pores.

### 3.1.2 Interaction energy in silica pores

To quantify the fundamental mechanism responsible for the results summarised in **Table 3**, we calculated the interaction energy between selected fluid molecules and the silica substrate. For the *n*-butane-surface interaction energy, we calculated the corresponding LJ potential. The Coulombic potential is not considered as *n*-butane is not charged in our model. Plots of interaction energy versus

simulation time for all fluid molecules over the last 10ns of the simulations are presented in Figure S2 of the **SI** and show that the interaction energies of the systems are stable. The *n*-butane-surface interaction energy, normalized by the number of *n*-butane is shown in **Table 4**. The results are consistent with an attraction between *n*-butane and the pores. As H<sub>2</sub>S or CO<sub>2</sub> are added to the system, the *n*-butane – pore interaction energy becomes less attractive. This result is qualitatively consistent with the percentage displaced of *n*-butane shown in **Table 3**, as higher percentage displaced correlates with less attractive interaction energy.

We also determined the CO<sub>2</sub>-pore and H<sub>2</sub>S-pore interaction energies ( $E_{\text{CO}_2\text{-pore}}$  and  $E_{\text{H}_2\text{S-pore}}$ ). For these calculations, we considered both LJ and electrostatic contributions to the interaction energy. The results are normalised by the amount of CO<sub>2</sub> or H<sub>2</sub>S in the system. The results, presented in **Table 5** show that CO<sub>2</sub> is more strongly attracted to the pore surfaces than H<sub>2</sub>S. This difference could explain why CO<sub>2</sub> is more effective than H<sub>2</sub>S at reducing *n*-butane density near the silica pore surface. The results also indicate that the normalised interaction energy is more attractive when fewer acid gas molecules are present in the system. This is consistent with previous reports for CO<sub>2</sub>-CH<sub>4</sub> confined in calcite<sup>25</sup> and in silica pores,<sup>57</sup> and suggests that the first acid gas molecules in the system adsorb on the preferential adsorption sites, possibly via electrostatic interactions related to the formation of hydrogen bonds. For the specific conditions considered, the preferential adsorption sites are expected to be the –OH groups on the surface. Analysis of our simulations suggests that 232 CO<sub>2</sub> and 238 H<sub>2</sub>S molecules saturate each of the 2 pore surfaces available in our simulations. Density profiles in **Figure 3** show that some molecules of both CO<sub>2</sub> and H<sub>2</sub>S occupy the region close to the centre of the pore even before the surface becomes saturated, suggesting some exchange between adsorbed and pore gases.

**Table 4:** *n*-Butane interaction energy with silica. All systems contain 300 *n*-butane molecules

Acid gas loading	With H <sub>2</sub> S (kJ/mol)	With CO <sub>2</sub> (kJ/mol)
200	-3.13±0.01	-3.10±0.01
375	-2.65±0.02	-2.51±0.02
500	-2.37±0.02	-2.23±0.01
Pure <i>n</i> -butane	-3.74±0.01	

**Table 5:** Gas-silica pore interaction energy for H<sub>2</sub>S and CO<sub>2</sub>.

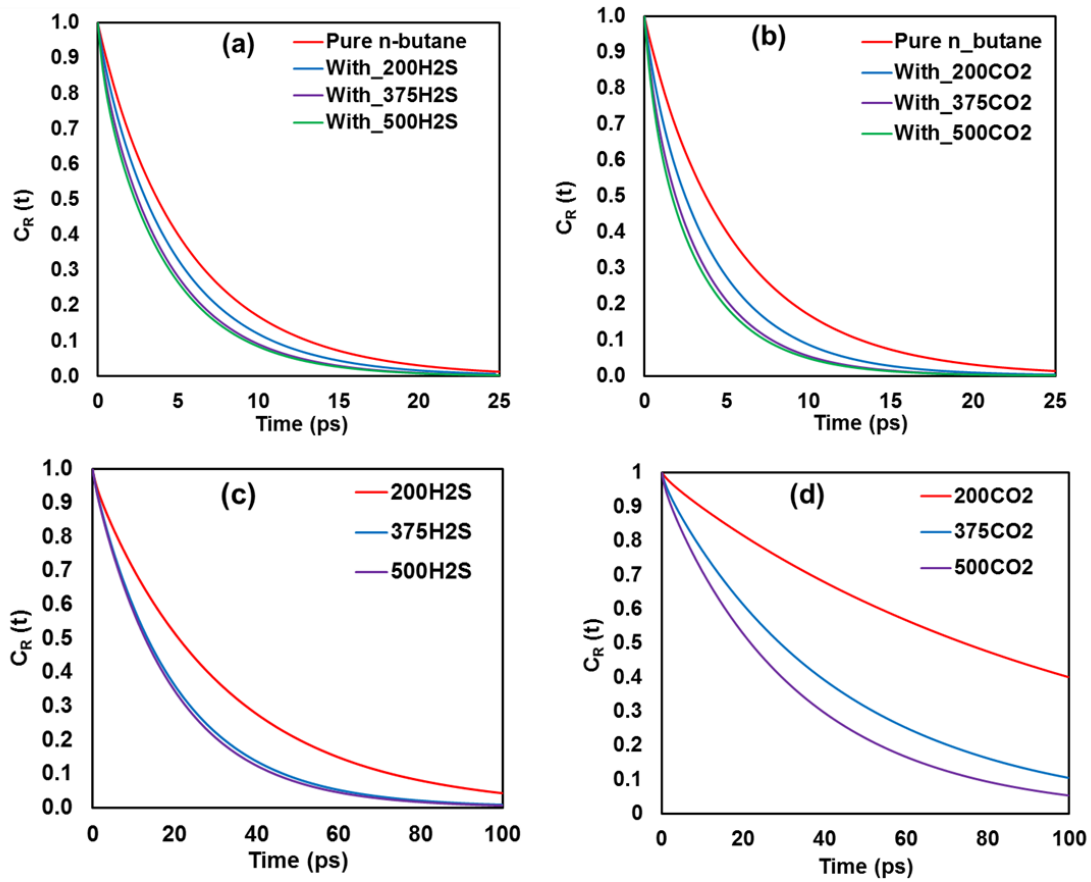
Gas Loading	E <sub>H<sub>2</sub>S-Surf</sub> (Electrostatic)(kJ/mol)	E <sub>H<sub>2</sub>S-Surf</sub> (LJ) (kJ/mol)	E <sub>CO<sub>2</sub>-Surf</sub> (Electrostatic)(kJ/mol)	E <sub>CO<sub>2</sub>-Surf</sub> (LJ)(kJ/mol)
200	-10.63±0.09	-2.77±0.01	-18.0±0.1	-4.05±0.02
375	-9.31±0.05	-2.57±0.01	-13.5±0.1	-3.35±0.01
500	-8.54±0.03	-2.42±0.01	-11.2±0.2	-2.94±0.01

### 3.1.3 Residence times

To further characterise the systems considered in **Figure 2**, we calculated the residence time via the residence correlation function (RCF),  $C_R(t)$ , for *n*-butane molecules found within a distance of 5Å from the pore surface, which belongs to the first adsorbed layer. The  $C_R(t)$  is defined as:<sup>37</sup>

$$C_R = \frac{\langle N_i(t)N_i(0) \rangle}{\langle N_i(0)N_i(0) \rangle} \quad (2)$$

In Eq. 2,  $N_i(t) = 1$  if molecule *i* resides in the layer considered at time *t*, and 0 otherwise.  $N_i(0) = 1$  if molecule *i* belongs to the layer at time  $t = 0$  and becomes 0 only when molecule *i* leaves the layer and remains equal to zero even though the molecule returns to the layer. The faster  $C_R(t)$  decays from 1 to 0, the faster molecules leave the layer considered, which in our case is the FAL. The COM of *n*-butane is used to represent the position of one *n*-butane molecule in this calculation. The  $C_R(t)$  calculated for *n*-butane as a function of H<sub>2</sub>S and CO<sub>2</sub> loading are shown in **Figure 4**. The results show that in both *n*-butane-H<sub>2</sub>S and *n*-butane-CO<sub>2</sub> systems, the autocorrelation functions decay to zero faster as the gas loading increases. Comparing results obtained for H<sub>2</sub>S with CO<sub>2</sub> at the same loading shows faster decays in the presence of CO<sub>2</sub>. This suggests that the interactions between CO<sub>2</sub> and the pore surface weaken the surface-*n*-butane interactions more effectively than those between H<sub>2</sub>S and the surface. The  $C_R(t)$  for H<sub>2</sub>S and CO<sub>2</sub> molecules in the first adsorbed layer are shown in **Figures 4c** and **4d**. We considered the C of CO<sub>2</sub> and the S of H<sub>2</sub>S for the  $C_R(t)$  calculation. The results show that the residence times of gas molecules in the first adsorbed layer decreases as loading increases. This is probably due to fast exchange between gas molecules in the adsorbed layer and those close to the centre of the pore. This observation is consistent with previous reports in literature,<sup>24</sup> with our density profiles, and also with the interaction energies discussed above.



**Figure 4:** Residence CF for  $n$ -butane molecules in the FAL at different  $H_2S$  (a) and  $CO_2$  loading (b). The CF for system containing pure  $n$ -butane is also shown for comparison. CF of  $H_2S$  molecules in the FAL at different loading are shown in (c) and for  $CO_2$  molecules at different loading in (d).

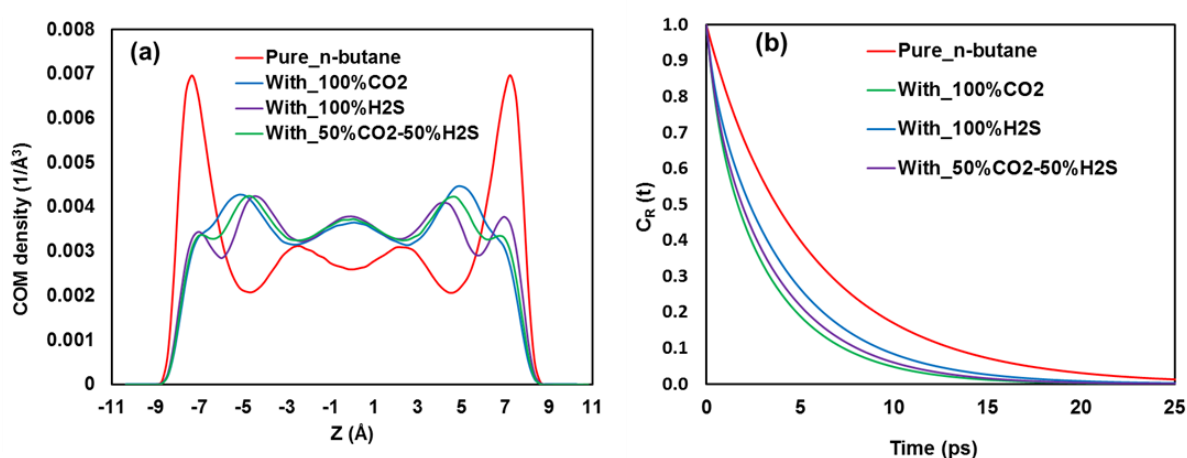
## 3.2 Ternary Systems

### 3.2.1 Effect of $H_2S$ - $CO_2$ mixtures on $n$ -butane displacement

To investigate whether mixtures of  $H_2S$  and  $CO_2$  could act synergistically in terms of  $n$ -butane displacement from the pore surfaces, we quantified the effect of 50%-50% mixtures of  $CO_2$  and  $H_2S$  in the presence of  $n$ -butane (system 10 in **Table 1**). The corresponding  $n$ -butane density profiles are compared in **Figure 5a**. The results show that the distribution of  $n$ -butane in the presence of the  $H_2S$  -  $CO_2$  mixture is between those obtained when either  $CO_2$  or  $H_2S$  is present, yielding a percentage displaced of  $\sim 62\%$ . These results suggest that mixing these gases is not synergistic, but rather reduces the ability of  $CO_2$  to displace  $n$ -butane from the pore surfaces by  $\sim 5\%$ . This is consistent with the report of Khan et al.,<sup>34</sup> who used reservoir simulations to quantify the effect of acid gas injection on

enhanced gas recovery. Results for the *n*-butane  $C_R(t)$ , shown in **Figure 5b**, are also consistent with what has just been discussed.

To further understand how  $H_2S$ - $CO_2$  mixtures could displace *n*-butane from pore surfaces, we simulated additional ternary systems varying the gas composition while maintaining a constant acid gas loading of 375 molecules. We simulated  $CO_2:H_2S$  ratios of 1:2 and 2:1 (systems 11 and 12 in **Table 1**). The density profiles, presented in **SI**, suggest that the higher the  $CO_2$  fraction in the gas mixture, the better the displacing performance.

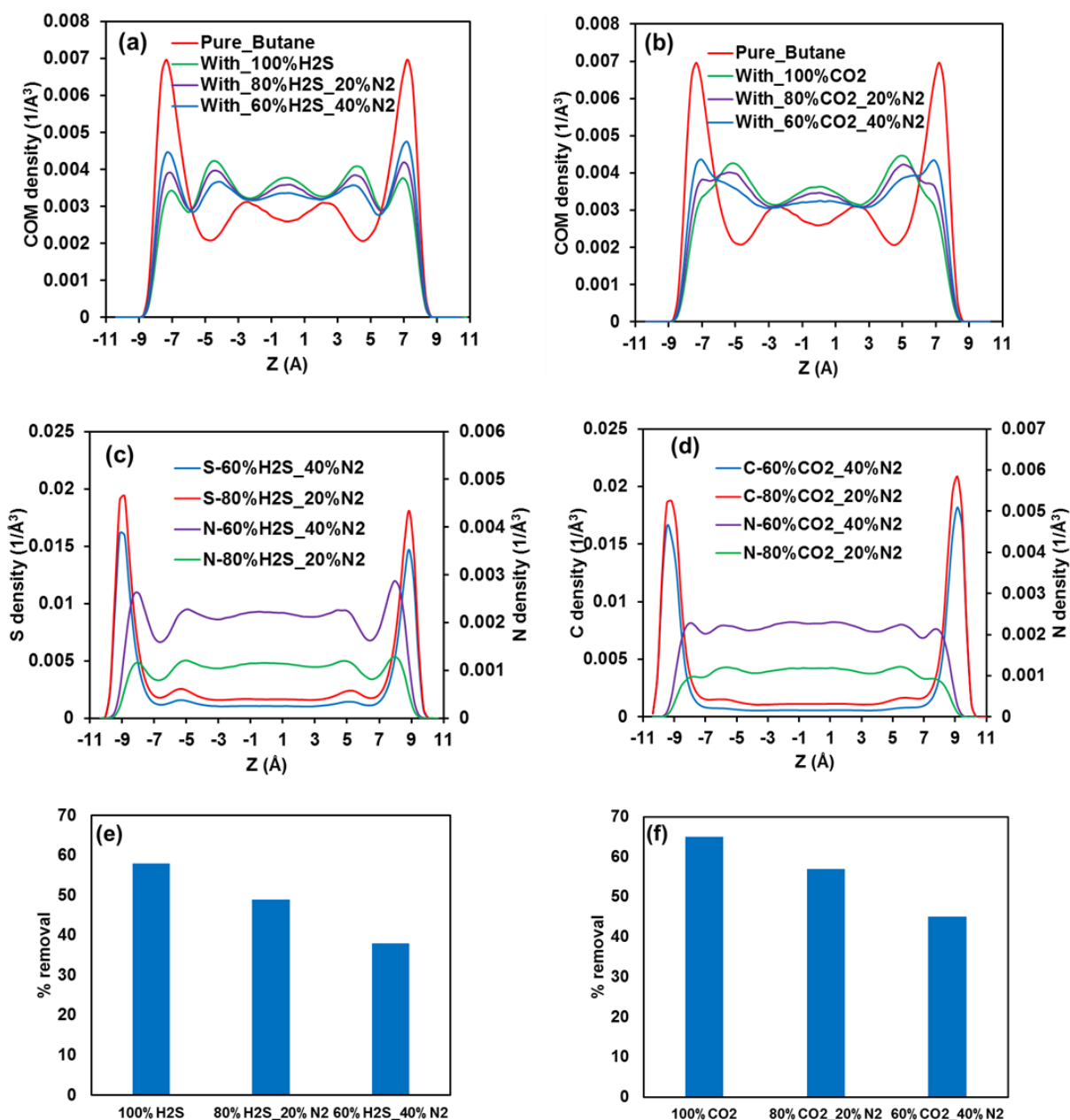


**Figure 5:** Comparison of density profiles of *n*-butane in the presence of pure  $CO_2$  and  $H_2S$  and their mixture (a) and residence CF for the same systems (b). The binary systems contain 300 *n*-butane and 500 molecules of the gases. The mixture comprises 250 molecules for each of the acid gas.

### 3.2.2 Effect of $CO_2/H_2S/N_2$ mixture composition on *n*-butane displacement

The results discussed so far suggest that the ability of  $H_2S$  and  $CO_2$  to displace hydrocarbons from the pore surface is related to their strong pore-wall attractions, sometimes due to electrostatic interactions. To further test this observation, we conducted additional simulations in which nitrogen was added to various systems. We conducted simulations for 80% $CO_2$ -20% $N_2$  and 60% $CO_2$ -40% $N_2$  for a total gas loading of 500 molecules (systems 15 and 16 in **Table 1**), as well as for systems containing  $H_2S$  and  $N_2$  (systems 13 and 14 in **Table 1**). The results are shown in **Figure 6**. The density profiles for COM of *n*-butane are shown in **Figure 6a** and **6b** and those for C of  $CO_2$ , S of  $H_2S$  and N of  $N_2$  in **Figure 6c** and **6d**. The estimated percentage displaced for the various systems is shown in **Figure 6e** and **6f**. The results show that as the  $N_2$  mole fraction in the mixture increases, the percentage displaced decreases.





**Figure 6:** Density profiles for COM of *n*-butane for (a) H<sub>2</sub>S-N<sub>2</sub> system (b) CO<sub>2</sub>-N<sub>2</sub> system. Density profiles for N of N<sub>2</sub> for (c) H<sub>2</sub>S-N<sub>2</sub> (d) CO<sub>2</sub>-N<sub>2</sub> systems. There are 300 *n*-butane molecules in all cases. The total number of gas molecules is 500. Percentage of hydrocarbon displaced is also shown for H<sub>2</sub>S-N<sub>2</sub> (e) and CO<sub>2</sub>-N<sub>2</sub> systems (f).

### 3.3 Effect of Pore Surface Chemistry

#### 3.3.1 Density profiles and percentage displaced

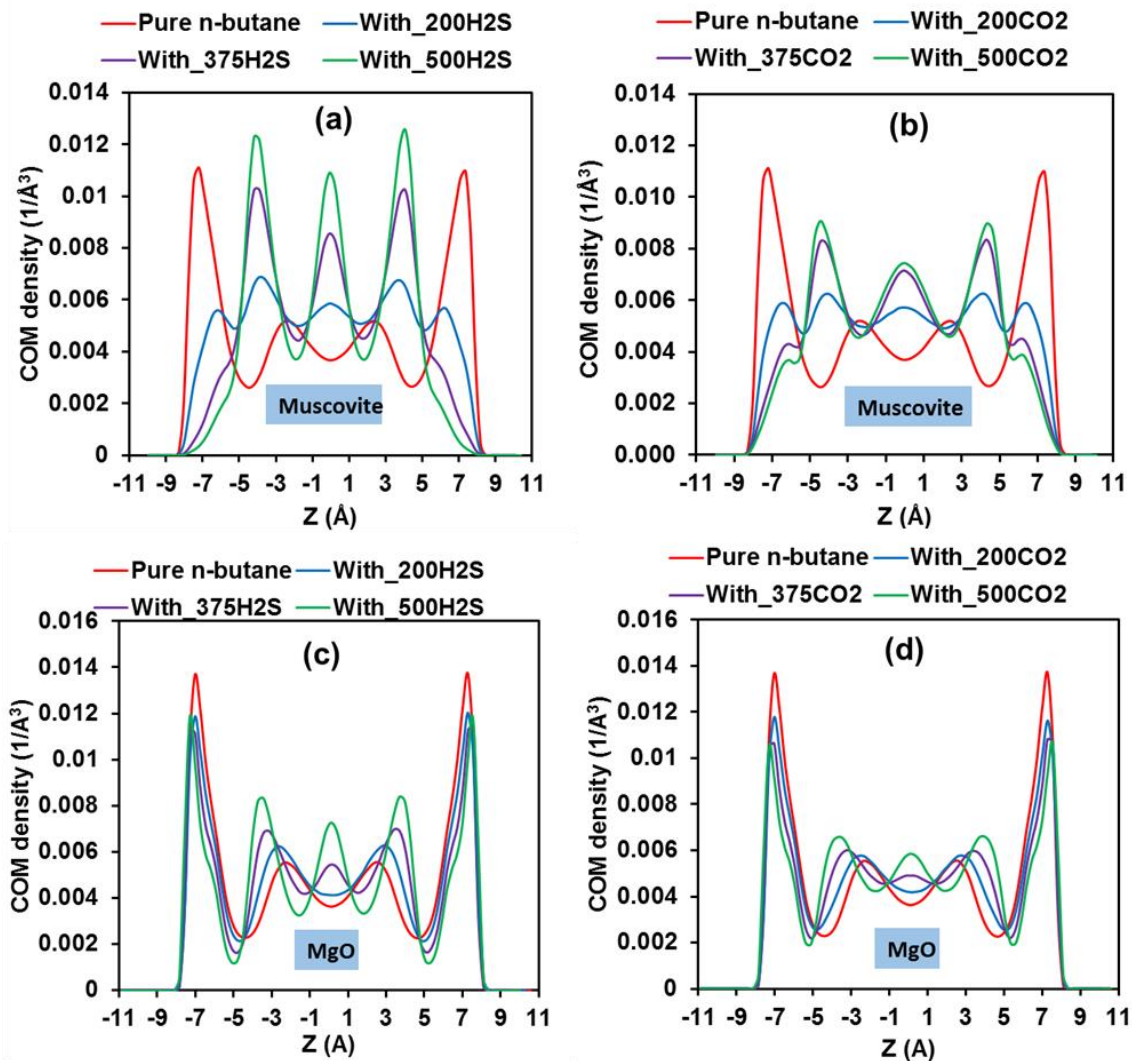
To investigate whether the performance of the acid gases depends on the nature of the substrate, we performed additional simulations in muscovite and magnesium oxide (MgO) pores. The corresponding system compositions are shown in **Table 1**. The CO<sub>2</sub> and H<sub>2</sub>S gases were added into pores containing 300 *n*-butane molecules. The number of *n*-butane molecules is constant for all systems. It should be noted that the bulk pressures corresponding to systems with H<sub>2</sub>S and CO<sub>2</sub> are similar (see **Table 2**). The density profiles for the COM of *n*-butane at different loadings are shown in **Figure 7**. The calculated percentage *n*-butane displaced from the pore surfaces are presented in **Figure 8**. The results show (a) higher percentage displaced for H<sub>2</sub>S than CO<sub>2</sub> in muscovite; (b) both gases yield similar, rather poor performance in MgO. It should be noted that the pressures considered in the silica substrate (see **Table 2**) are somewhat different compared to those considered in the other two substrates. However, because the results are due to preferential fluid-surface interactions, we do not expect this difference to affect qualitatively the results presented. These results are certainly dependent on the protonation states of the surfaces considered here, as these affect the interactions between the various fluid molecules and the pore surfaces. Because water was not considered in our systems, we did not explicitly quantify the effect of varying surface protonation on the results presented. The density profiles of carbon and oxygen of CO<sub>2</sub>, sulphur and hydrogens of H<sub>2</sub>S and N of N<sub>2</sub> for binary systems in silica and MgO pores are shown in **SI**. In all cases, CO<sub>2</sub> and H<sub>2</sub>S are preferentially adsorbed on the surfaces.

Results for *n*-butane-pore interaction energies (**Figure 9**) show that both CO<sub>2</sub> and H<sub>2</sub>S reduce *n*-butane – pore attraction except in MgO, in which case the interaction energy remains relatively unchanged upon gas loading. The results for the normalised interaction energies between the gases and the substrates, shown in **Figure 10**, show that, while CO<sub>2</sub> is more strongly attracted to silica than H<sub>2</sub>S, H<sub>2</sub>S is more strongly attracted to muscovite than CO<sub>2</sub>. Both gases have similar interaction with MgO. These results suggest that the more strongly a gas is attracted to a porous substrate, the more effective it is expected to be at displacing hydrocarbons from the surface of the corresponding pore.

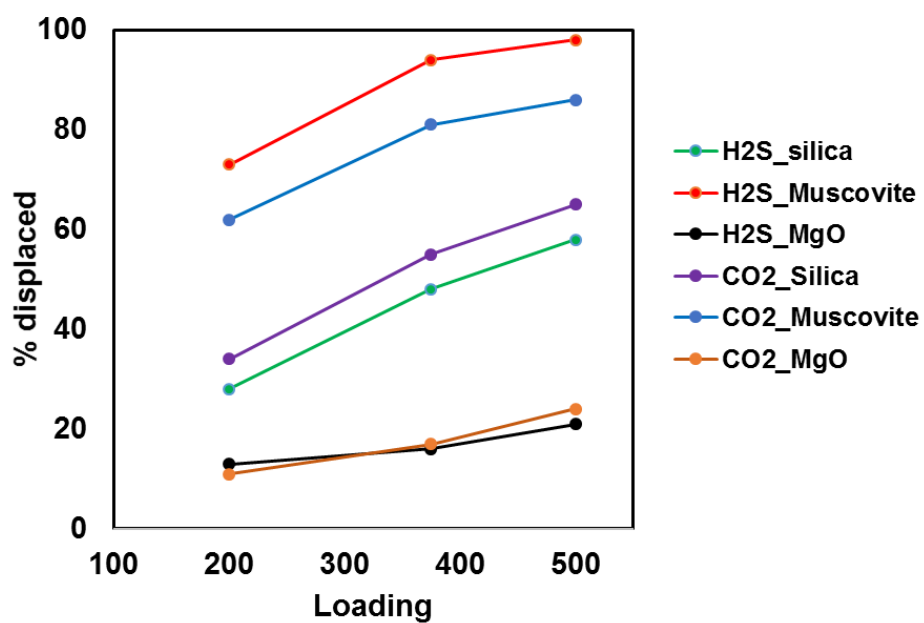
Results for *n*-butane residence correlation function, C<sub>R</sub>(t) are described in **SI**, and indicate that: (a) in silica, the residence time of *n*-butane decreases with H<sub>2</sub>S and CO<sub>2</sub> loadings; (b) in muscovite, C<sub>R</sub>(t)

essentially overlap as CO<sub>2</sub> loading changes, suggesting that *n*-butane exchange between adsorbed molecules close to the pore walls and to the pore centre occurs with similar frequency irrespective of CO<sub>2</sub> loading; (c) for H<sub>2</sub>S loading in muscovite, residence time decreases with loading; and (d) in MgO pores, C<sub>R</sub>(t) increases with increase in H<sub>2</sub>S and CO<sub>2</sub> loading.

In general, the results just discussed suggest that, for dry systems, it is possible to identify those gases that can more effectively displace short linear hydrocarbons from the pore surfaces by considering the interactions of such gases with the solid themselves: the gases that are more strongly attracted to the pore surfaces are likely to be more effective at displacing the hydrocarbons. Because realistic materials are heterogeneous, lab-scale experimental tests should be conducted before field campaigns.



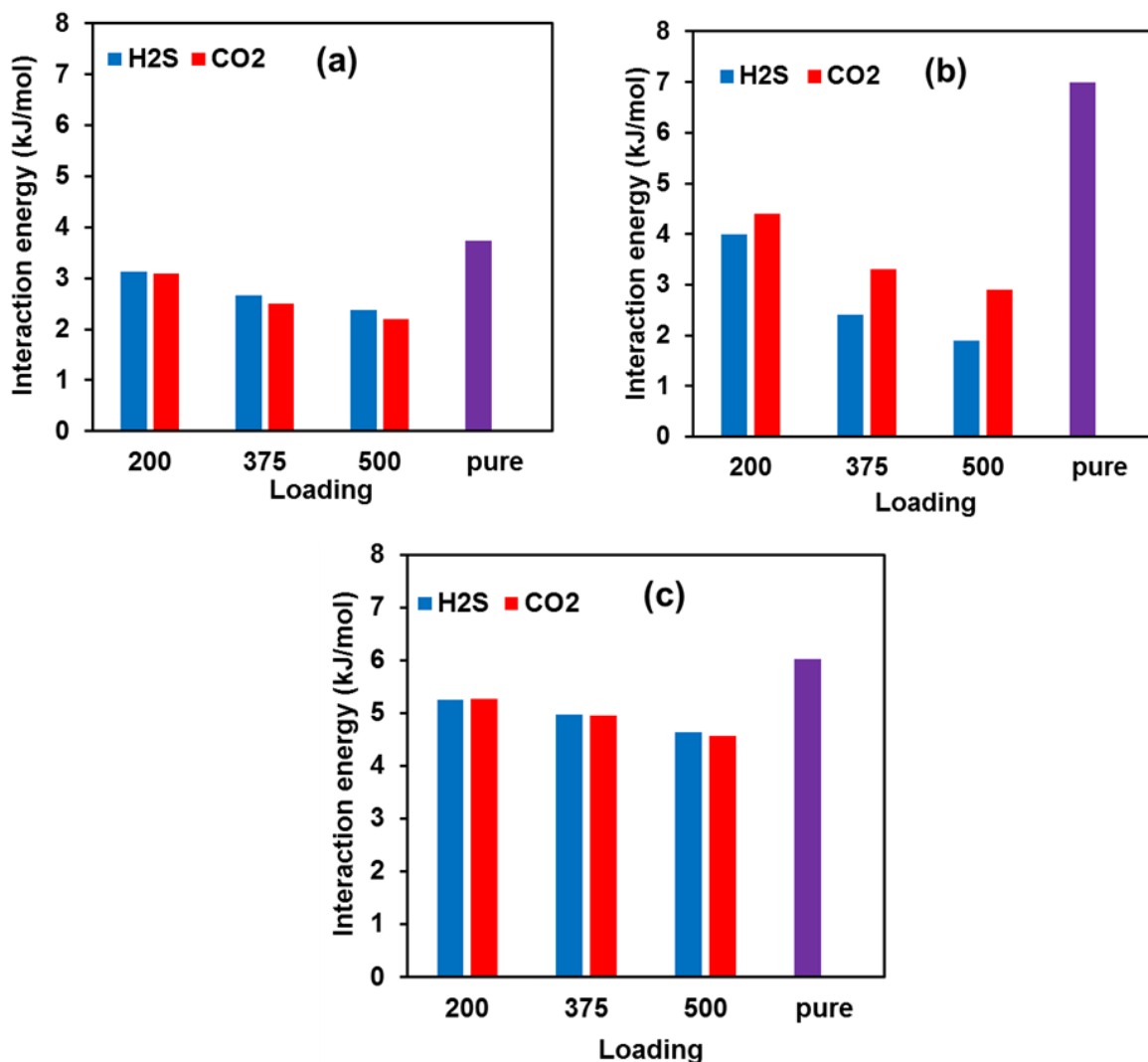
**Figure 7:** *n*-Butane density profiles at different gas loadings for H<sub>2</sub>S in muscovite (a) CO<sub>2</sub> in muscovite (b) H<sub>2</sub>S in MgO (c) and CO<sub>2</sub> in MgO (d). The system comprises 300 *n*-butane molecules in all cases.



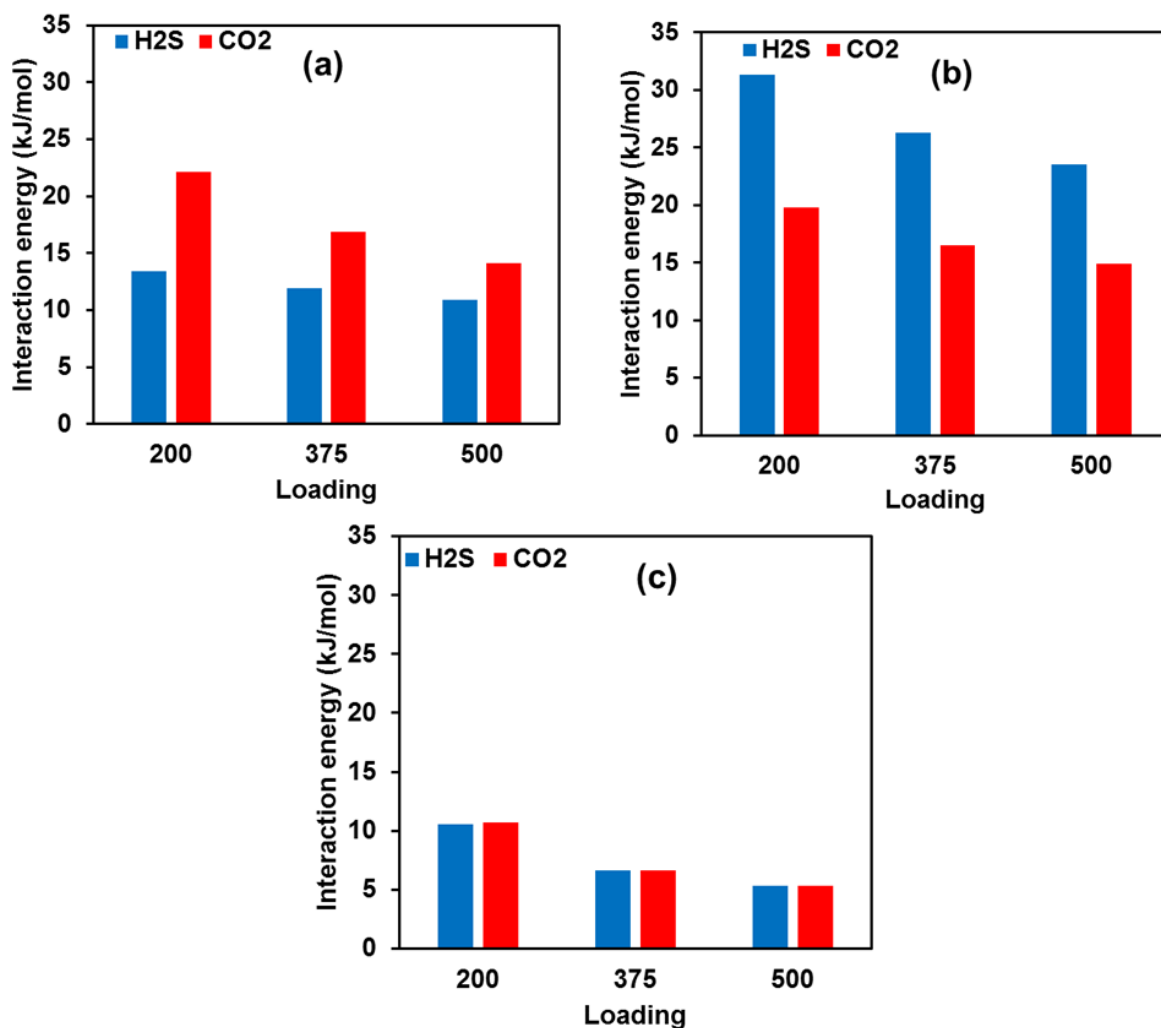
**Figure 8:** Percentage of *n*-butane displaced by H<sub>2</sub>S and by CO<sub>2</sub> in the substrates considered.

### 3.3.3 Effect of nitrogen in muscovite and MgO pores

Additional simulations were conducted in which nitrogen was added to the confined systems instead of the other gases. The results, shown in **SI**, are consistent with those discussed for silica pores and suggest that N<sub>2</sub> has little or no effect on *n*-butane displacement from the pore surfaces even when muscovite or MgO pores are considered. In all cases, nitrogen seems to preferentially distribute near the pore centre.



**Figure 9:** Magnitude of total interaction energy of *n*-butane at different gas loading for silica (a) muscovite (b) and MgO (c). The systems comprise 300 *n*-butane molecules and various amounts of the other fluids. The interaction energies are normalised by the number of *n*-butane molecules in the system. The interaction energy shown is only due to van der Waals interaction as *n*-butane molecules bear no charge.



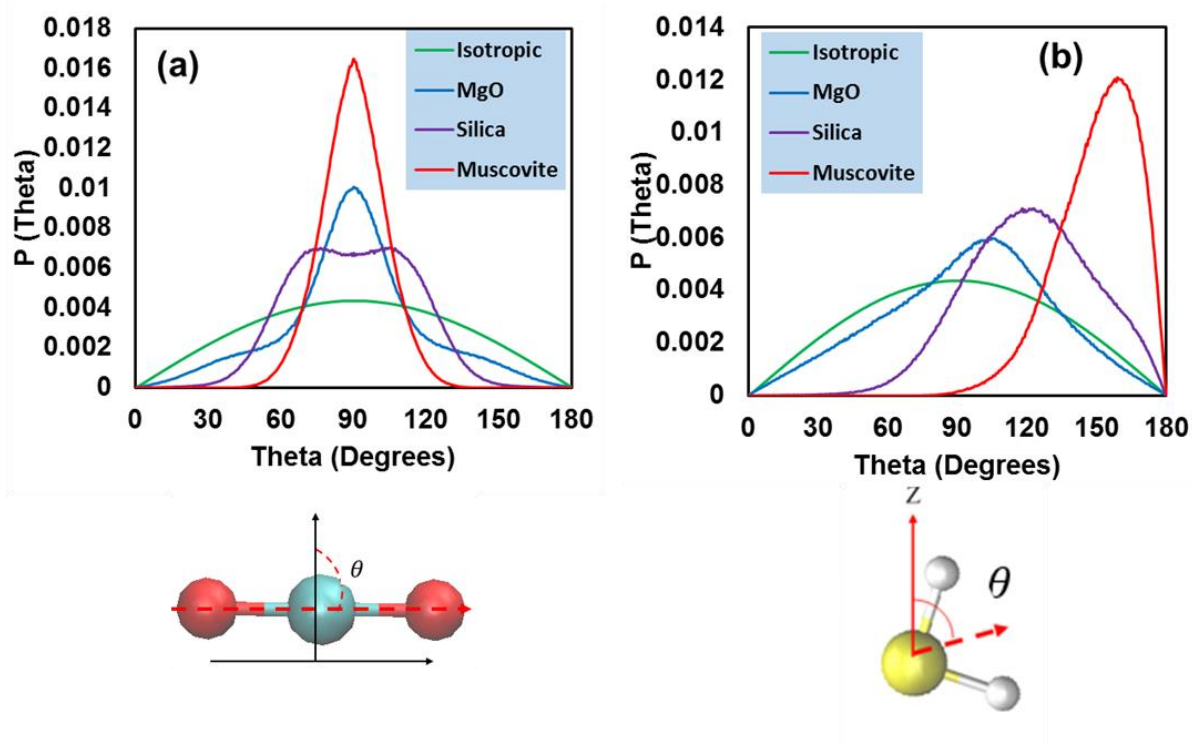
**Figure 10:** Magnitude of total interaction energy of CO<sub>2</sub> and H<sub>2</sub>S at different loading for silica (a) muscovite (b) and MgO (c). The systems comprise 300 *n*-butane molecules in all cases. The interaction energies are normalised by the number of gas molecules in the system. The interaction energy shown is the sum of the LJ and electrostatic contributions.

### 3.4 Molecular Structure of Adsorbed Gases

The results above demonstrate that different gases can have various effects in displacing short linear hydrocarbons from pore surfaces because of their different interactions with the pore surfaces. Because the interaction energies depend on the various chemical species found on the pore surface, it is of fundamental interest to quantify how the various fluid molecules assemble within a pore, with relevance to the distribution of the atomic species on the solid substrate. We consider here gas molecules adsorbed within the FAL, because of the relevance of the FAL, based on our prior results, on determining the distribution and the diffusion coefficients of various fluid molecules within narrow pores.<sup>24</sup>

### 3.4.1 Orientation of adsorbed gases in the first adsorbed layer

We provide the preferential orientation of CO<sub>2</sub> and H<sub>2</sub>S molecules in contact with all simulated substrates in **Figure 11a** and **11b**, respectively. The molecules considered in the calculation are those found within 2 Å from silica and muscovite surfaces and 5 Å from MgO surface. The results are obtained at the maximum loading conditions. The thickness of the adsorbed layers considered (i.e., 2 vs. 5 Å), is chosen based on the features of the density profiles, to focus on the FAL. The orientation of CO<sub>2</sub> molecules were quantified in terms of the distribution of the angle formed between the CO<sub>2</sub> backbone and the surface normal. If the angle equals 0° or 180°, CO<sub>2</sub> is perpendicular to the surface, when it is 90°, CO<sub>2</sub> lays parallel to the surface. For H<sub>2</sub>S, we plot the distribution of the angle formed between the vector pointing from S to the midpoint of H-H vector and the surface normal. When the angle equals 0°, H<sub>2</sub>S points the two hydrogen atoms away from the surface, and when it is 180°, H<sub>2</sub>S points both hydrogens towards the surface. When theta is 90°, H<sub>2</sub>S either points one hydrogen to the surface or lays parallel to the surface. **Figure 11a** and **11b** show that the gases adopt different orientation on each of the substrates. On silica, CO<sub>2</sub> yields an angle of ~ 75° to the surface. A similar orientation of CO<sub>2</sub> has been previously reported for silica.<sup>23</sup> For muscovite and MgO, in the models considered here, CO<sub>2</sub> lays parallel to the surface. On the other hand, H<sub>2</sub>S on muscovite is almost in 2-hydrogen down orientation, while on silica and MgO its orientation seems to be with either one hydrogen towards the surface or parallel to it. The results show that the nature of the substrates dictates the structure of adsorbed gas, which could influence the EOR performance of the gas. For example, H<sub>2</sub>S tends to perform better than CO<sub>2</sub> in muscovite where it almost points its two hydrogen atoms towards the surface when compared to its performance in silica where it preferentially points one hydrogen down or lays parallel to the surface. Note that CO<sub>2</sub> performs better than H<sub>2</sub>S in silica. This suggests that H<sub>2</sub>S could be effective at displacing hydrocarbons from substrates in which active sites are available on the solid surfaces where both hydrogen atoms of H<sub>2</sub>S could be strongly attracted.



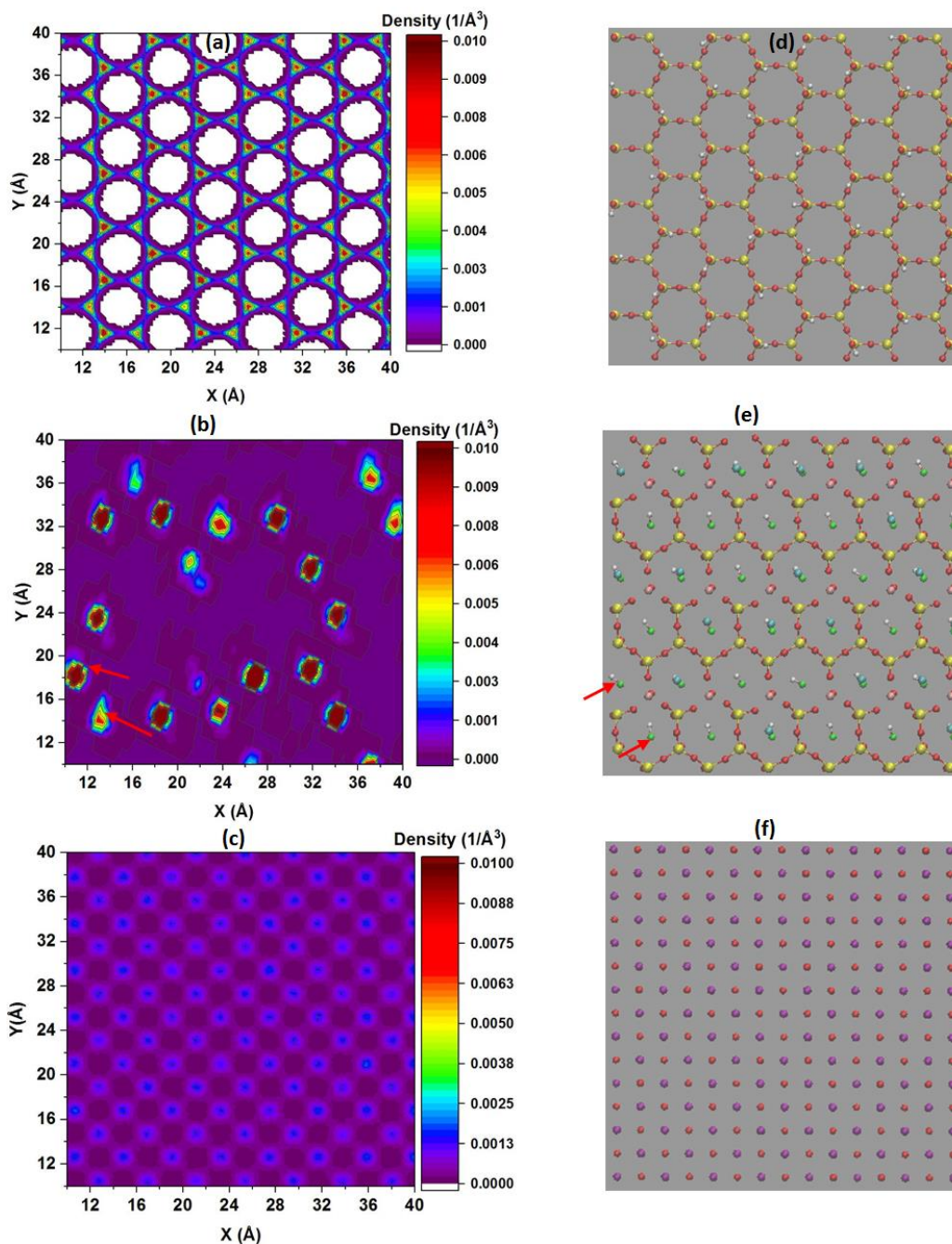
**Figure 11:** (a) Probability distribution of angle theta for CO<sub>2</sub> molecules (a) and H<sub>2</sub>S molecules (b) adsorbed within 2Å from silica and muscovite surfaces and 5Å from MgO surface. The results are compared with the isotropic distribution (green). All systems contain 300 *n*-butane and 500 acid gas molecules.

### 3.4.2 Planar density distributions

To further quantify how the solid substrates determine the distribution of H<sub>2</sub>S within the FAL, we study the in-plane density distributions of H<sub>2</sub>S and we relate it to the atomic distribution on the three surfaces. In **Figure 12**, we present the in-plane density distribution of sulphur (S) of H<sub>2</sub>S in all simulated substrates. The correspondent results for the hydrogen atoms (H) of H<sub>2</sub>S are shown in **SI**. The results reveal the preferential adsorption sites for H<sub>2</sub>S. For example, on silica, S of H<sub>2</sub>S preferentially interacts with the vertices of the hexagonal rings formed by the silicon atoms of the substrate. In muscovite, S of H<sub>2</sub>S preferentially adsorbs on the hydroxyl groups in muscovite interlayer. On the other hand, for MgO, S of H<sub>2</sub>S adsorbs near the oxygen atoms.

Similar results are shown in **SI** for CO<sub>2</sub>.





**Figure 12:** In-plane density distribution of S of H<sub>2</sub>S at contact with (a) silica (b) muscovite (c) MgO. Snapshot of silica (d) muscovite (e) and MgO (f) are also shown. All systems contains 500 H<sub>2</sub>S and 300 *n*-butane. Colour code: OH oxygen - green , bridging oxygen – red, silicon – yellow, potassium – cyan, purple – magnesium, hydrogen – white.

### 3.5 Diffusion Coefficients of Confined Fluids

To quantify the diffusion coefficients, we followed established procedures, starting from the mean square displacement (MSD) of the COM of *n*-butane, acid gases, and N<sub>2</sub> as a function of time (shown in Supporting Information).<sup>58</sup> The diffusion coefficient results presented in **Tables 6-8** show that CO<sub>2</sub>,

H<sub>2</sub>S and N<sub>2</sub>, in general, slow down the diffusion of *n*-butane in all simulated substrates, probably because of pore crowding. An exception is in the case of the lowest CO<sub>2</sub> loading (200 CO<sub>2</sub>) in silica, wherein the self diffusion coefficient obtained for *n*-butane is slightly higher than that observed for the pure *n*-butane when no CO<sub>2</sub> is present. To check whether CO<sub>2</sub> could enhance the *n*-butane diffusion at lower loading (“molecular lubrication”), we simulated additional systems where 300 *n*-butane molecules were simulated together with either 125 or and 75 CO<sub>2</sub> molecules. We obtained  $6.4 \times 10^{-8}$  and  $6.5 \times 10^{-8}$  m<sup>2</sup>/s for *n*-butane diffusion coefficient when 75 and 125 CO<sub>2</sub> molecules were present within the silica pore, respectively. These results are similar to those obtained for pure *n*-butane, suggesting that CO<sub>2</sub> could enhance *n*-butane transport, but that pore crowding quickly suppresses this effect, at the conditions considered in this report. For completeness, the self-diffusion coefficients obtained for CO<sub>2</sub> when either 75 or 125 CO<sub>2</sub> molecules were present together with 300 molecules of *n*-butane were found to be  $2.7 \times 10^{-9}$  and  $3.9 \times 10^{-9}$  m<sup>2</sup>/s, respectively.

**Table 6:** Self-diffusion coefficient for *n*-butane, CO<sub>2</sub>, H<sub>2</sub>S, and N<sub>2</sub> in silica pores. Please refer to **Table 1** for system composition.

System	D( <i>n</i> -butane) (10 <sup>-8</sup> m <sup>2</sup> /s)	D(H <sub>2</sub> S) (10 <sup>-8</sup> m <sup>2</sup> /s)	D(CO <sub>2</sub> ) (10 <sup>-8</sup> m <sup>2</sup> /s)	D(N <sub>2</sub> ) (10 <sup>-8</sup> m <sup>2</sup> /s)
1	6.5±0.1	1.8±0.02	-	-
2	6.2±0.2	1.5±0.02	-	-
3	5.5±0.2	1.3±0.04	-	-
4	7.0±0.4	-	0.6±0.03	-
5	6.4±0.3	-	1.0±0.01	-
6	6.6±0.4	-	1.1±0.01	-
7	5.8±0.2	-	-	4.2±0.05
8	5.8±0.2	-	-	2.6±0.04
9	5.2±0.1	-	-	2.0±0.04
Pure <i>n</i> -butane	6.5±0.2	-	-	-

The results in **Tables 6-8** show that H<sub>2</sub>S and CO<sub>2</sub> have lower self-diffusion coefficients than *n*-butane, reflecting the stronger interaction with the substrates, as discussed above. The results suggest that H<sub>2</sub>S travels faster than CO<sub>2</sub> through the silica pores, consistent with the weaker attraction to this substrate. In muscovite pores, CO<sub>2</sub> travels faster than H<sub>2</sub>S while both acid gases has similar diffusion coefficient in MgO pores. These results are consistent with interaction energy results presented in **Figure 10**. Results in **Tables 6-8** also show that N<sub>2</sub> travels faster than H<sub>2</sub>S and CO<sub>2</sub> especially at low N<sub>2</sub> loading, perhaps due to smaller size of N<sub>2</sub> molecule and also reflecting the fact that N<sub>2</sub> does not show strong adsorption to the pore surfaces in the systems considered here.

Some prior studies reported an enhancement of the mobility of hydrocarbons within model pores upon low CO<sub>2</sub> loading.<sup>24, 26, 59</sup> This was not observed in the present simulations, because the systems

considered here are very dense, and pore crowding is expected to inhibit diffusion. In fact, prior studies have reported a decrease in hydrocarbon mobility at higher CO<sub>2</sub> loadings.<sup>59</sup>

**Table 7:** Diffusion coefficients for *n*-butane, CO<sub>2</sub>, H<sub>2</sub>S, and N<sub>2</sub> in muscovite pores. Please refer to **Table 1** for system composition.

System	D( <i>n</i> -butane) (10 <sup>-8</sup> m <sup>2</sup> /s)	D(H <sub>2</sub> S) (10 <sup>-9</sup> m <sup>2</sup> /s)	D(CO <sub>2</sub> ) (10 <sup>-9</sup> m <sup>2</sup> /s)	D(N <sub>2</sub> ) (10 <sup>-9</sup> m <sup>2</sup> /s)
1	3.7±0.1	3.7±0.1	-	-
2	3.5±0.3	3.1±0.1	-	-
3	2.9±0.2	1.9±0.1	-	-
4	3.8±0.3	-	4.8±0.1	-
5	3.5±0.2	-	3.4±0.1	-
6	2.9±0.1	-	2.5±0.1	-
7	0.9±0.2	-	-	5.3±0.3
8	0.9±0.1	-	-	2.6±0.2
9	0.7±0.1	-	-	1.0±0.2
Pure <i>n</i> -butane	4.1±0.2	-	-	-

**Table 8:** Diffusion coefficients for *n*-butane, CO<sub>2</sub>, H<sub>2</sub>S, and N<sub>2</sub> in MgO pores. Please refer to **Table 1** for system composition.

System	D( <i>n</i> -butane) (10 <sup>-8</sup> m <sup>2</sup> /s)	D(H <sub>2</sub> S) (10 <sup>-9</sup> m <sup>2</sup> /s)	D(CO <sub>2</sub> ) (10 <sup>-9</sup> m <sup>2</sup> /s)	D(N <sub>2</sub> ) (10 <sup>-9</sup> m <sup>2</sup> /s)
1	3.1±0.2	11.4±0.2	-	-
2	2.3±0.1	5.9±0.1	-	-
3	2.0±0.2	3.7±0.1	-	-
4	3.4±0.3	-	11.3±0.1	-
5	2.7±0.1	-	5.9±0.1	-
6	2.3±0.2	-	3.7±0.1	-
7	3.3±0.2	-	-	16.5±0.4
8	2.1±0.2	-	-	8.1±0.3
9	2.4±0.1	-	-	3.1±0.2
Pure <i>n</i> -butane	4.2±0.3	-	-	-

#### 4 CONCLUSIONS

Equilibrium MD simulations were performed to investigate the energetics, structure and transport properties of *n*-butane confined within slit-shaped nano-pores of width 2.2 nm carved out of silica, muscovite and MgO surfaces, in the presence of various amounts of CO<sub>2</sub>, H<sub>2</sub>S, N<sub>2</sub> and their mixtures at 350K. The study compares the effect of H<sub>2</sub>S vs. that of CO<sub>2</sub>, the effect of the presence of an inert gas (N<sub>2</sub>), and the effect of pore-surface chemistry on the behaviour of confined hydrocarbons. Because all

the pores considered were dry, pH effects on the pore surfaces were not investigated. Our results show that CO<sub>2</sub> is more efficient than H<sub>2</sub>S in displacing *n*-butane from silica surfaces, while H<sub>2</sub>S is more effective than CO<sub>2</sub> in muscovite. The two gases show similar performance in MgO. Analysis of the fluid-substrate interaction energy reveals that these results strongly correlate with the attraction between each acid gas and the pore surface. While H<sub>2</sub>S is more strongly attracted to muscovite compared to CO<sub>2</sub>, it is more weakly adsorbed on silica. Both gases show similar interaction with MgO. Our results also show that mixtures of CO<sub>2</sub> and H<sub>2</sub>S do not yield synergetic effects in displacing *n*-butane. The orientation and planar distribution of the adsorbed acid gases on the three solid surfaces further reinforce the observation that the nature of the pore surface dictates the structure of the adsorbed fluids, thus affecting how the acid gases control the structure of confined hydrocarbons. In most cases considered, our results show that the self-diffusion coefficient of *n*-butane decreases when other gases are added. This is probably due to a pore-crowding effect. Our results could contribute to the design of enhanced oil recovery strategies for improvement in hydrocarbon production and in acid gas sequestration. Future investigations should address the effect of water on the results presented.

## **SUPPORTING INFORMATION**

The Supporting Information is available free of charge on the ACS Publications website at DOI:

System composition determination; pressure calculation procedure; viscosity of bulk *n*-butane – N<sub>2</sub> systems; interaction energy as a function of time; simulation snapshots for *n*-butane – CO<sub>2</sub> and *n*-butane – N<sub>2</sub> systems; density profiles of *n*-butane –CH<sub>3</sub>, –CH<sub>2</sub> and COM in silica pores; density profiles of *n*-butane for gas mixtures at different ratios; density profiles of S of H<sub>2</sub>S, H of H<sub>2</sub>S, C of CO<sub>2</sub>, O of CO<sub>2</sub> and N of N<sub>2</sub> for binary systems of *n*-butane-acid gas and *n*-butane-N<sub>2</sub> in muscovite and MgO pores; density profiles of the centre of mass of *n*-butane, S of H<sub>2</sub>S, C of CO<sub>2</sub> and N of N<sub>2</sub> at different acid gas-N<sub>2</sub> ratios in the muscovite pores and MgO pores; we show the density profiles of *n*-butane for *n*-butane-N<sub>2</sub> systems in muscovite and MgO pores at different N<sub>2</sub> loadings. C<sub>R</sub> (t) for the COM of *n*-butane for binary systems of *n*-butane-acid gas in muscovite and MgO pores; in-plane density distributions for H of H<sub>2</sub>S within the FAL in silica, muscovite and MgO; in-plane density distribution of C and O of CO<sub>2</sub>

in contact with the surface for all simulated pores; mean square displacements for binary systems of *n*-butane – H<sub>2</sub>S, *n*-butane – CO<sub>2</sub> and *n*-butane – N<sub>2</sub> in silica, muscovite and MgO pores.

## **ACKNOWLEDGEMENTS**

Generous allocations of computing time were provided by the National Energy Research Scientific Computing Center (NERSC) at Lawrence Berkeley National Laboratory. NERSC is supported by the DOE Office of Science. S.B.B. acknowledges financial support from the Petroleum Technology Development Fund (PTDF). This research received funding from the European Union's Horizon 2020 research and innovation program under Grant No. 764810, within the consortium 'Science4CleanEnergy'. D.R.C. was supported by the U. S. Department of Energy, Office of Basic Energy Sciences, Division of Chemical Sciences, Geosciences and Biosciences under grant DE-SC0006878. We are grateful to Adrian Gregory and Prof. John Shaw for insightful discussions. Prof. Shaw's sabbatical at University College London was in part supported by the Leverhulme Trust under grant agreement number VP2-2017-023.

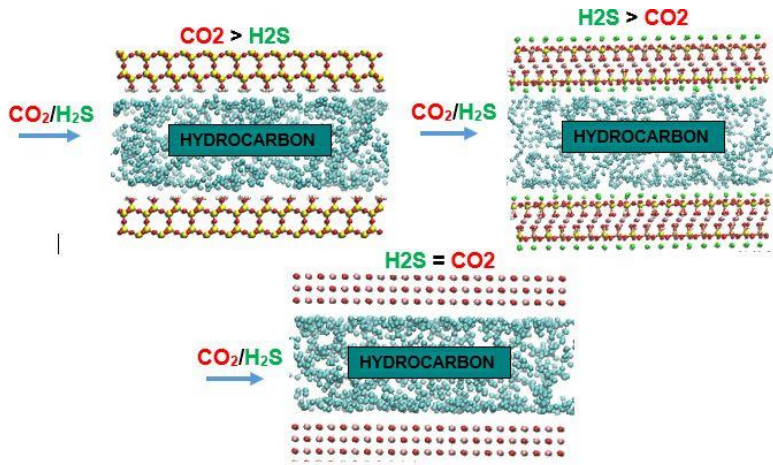
## REFERENCES

1. Malik, Q. M.; Islam, M. In *CO<sub>2</sub> Injection in the Weyburn Field of Canada: Optimization of Enhanced Oil Recovery and Greenhouse Gas Storage with Horizontal Wells*, SPE/DOE improved oil recovery symposium, Society of Petroleum Engineers: **2000**.
2. Ghaderi, S. M. Simulation Study of CO<sub>2</sub> Injection and Storage in Alberta. PhD Thesis, University of Calgary, **2013**.
3. Benson, S. M.; Deutch, J. Advancing Enhanced Oil Recovery as a Sequestration Asset. *Joule* **2018**, *2*, 1386-1389.
4. Niemi, A.; Bensabat, J.; Shtivelman, V.; Edlmann, K.; Gouze, P.; Luquot, L.; Hingerl, F.; Benson, S. M.; Pezard, P. A.; Rasmusson, K. Heletz Experimental Site Overview, Characterization and Data Analysis for CO<sub>2</sub> Injection and Geological Storage. *Int. J. GreenH Gas Con.* **2016**, *48*, 3-23.
5. Clark, D. E.; Gunnarsson, I.; Aradóttir, E.; Oelkers, E. H.; Sigfússon, B.; Snæbjörnsdóttir, S.; Matter, J. M.; Stute, M.; Júlíusson, B. M.; Gíslason, S. R. In *Mineral Storage of CO<sub>2</sub>/H<sub>2</sub>S Gas Mixture Injection in Basaltic Rocks*, AGU Fall Meeting Abstracts, 2017.
6. Orr, F. M.; Heller, J. P.; Taber, J. J. Carbon Dioxide Flooding for Enhanced Oil Recovery: Promise and Problems. *J. Am. Oil Chem. Soc.* **1982**, *59*, 810A-817A.
7. Fang, T.; Wang, M.; Gao, Y.; Zhang, Y.; Yan, Y.; Zhang, J. Enhanced Oil Recovery with CO<sub>2</sub>/N<sub>2</sub> Slug in Low Permeability Reservoir: Molecular Dynamics Simulation. *Chem. Eng. Sci.* **2018**.
8. Yan, Y.; Dong, Z.; Zhang, Y.; Wang, P.; Fang, T.; Zhang, J. CO<sub>2</sub> Activating Hydrocarbon Transport across Nanopore Throat: Insights from Molecular Dynamics Simulation. *Phys. Chem. Chem. Phys.* **2017**, *19*, 30439-30444.
9. Gunnarsson, I.; Aradóttir, E. S.; Oelkers, E. H.; Clark, D. E.; Arnarson, M. Þ.; Sigfússon, B.; Snæbjörnsdóttir, S. Ó.; Matter, J. M.; Stute, M.; Júlíusson, B. M. The Rapid and Cost-Effective Capture and Subsurface Mineral Storage of Carbon and Sulfur at the Carbfix2 Site. *Int. J. GreenH Gas Con.* **2018**, *79*, 117-126.
10. Cole, D. R.; Striolo, A. The Influence of Nanoporosity on the Behavior of Carbon-Bearing Fluids. Whole Earth Carbon book chapter Carbon Reactivity: 2019.
11. Yuan, S.; Wang, S.; Wang, X.; Guo, M.; Wang, Y.; Wang, D. Molecular Dynamics Simulation of Oil Detachment from Calcite Surface in Aqueous Surfactant Solution. *Comput. Theor. Chem* **2016**, *1092*, 82-89.
12. Jessen, K.; Tang, G.-Q.; Kovscek, A. R. Laboratory and Simulation Investigation of Enhanced Coalbed Methane Recovery by Gas Injection. *Transport Porous Med.* **2008**, *73*, 141-159.
13. Kowalczyk, P.; Gauden, P. A.; Terzyk, A. P.; Furmaniak, S.; Harris, P. J. Displacement of Methane by Coadsorbed Carbon Dioxide Is Facilitated in Narrow Carbon Nanopores. *J. Phys. Chem. C* **2012**, *116*, 13640-13649.
14. Yan, Y.; Li, C.; Dong, Z.; Fang, T.; Sun, B.; Zhang, J. Enhanced Oil Recovery Mechanism of CO<sub>2</sub> Water-Alternating-Gas Injection in Silica Nanochannel. *Fuel* **2017**, *190*, 253-259.
15. Sheng, J. J.; Chen, K. Evaluation of the Eor Potential of Gas and Water Injection in Shale Oil Reservoirs. *J. Unconv. Oil and Gas Res.* **2014**, *5*, 1-9.
16. Sheng, J. J. Critical Review of Field Eor Projects in Shale and Tight Reservoirs. *J. Petrol. Sci. and Eng.* **2017**, *159*, 654-665.
17. Sun, H.; Zhao, H.; Qi, N.; Li, Y. Effects of Surface Composition on the Microbehaviors of CH<sub>4</sub> and CO<sub>2</sub> in Slit-Nanopores: A Simulation Exploration. *ACS Omega* **2017**, *2*, 7600-7608.
18. Wang, S.; Javadpour, F.; Feng, Q. Fast Mass Transport of Oil and Supercritical Carbon Dioxide through Organic Nanopores in Shale. *Fuel* **2016**, *181*, 741-758.
19. Zhang, H.; Cao, D. Molecular Simulation of Displacement of Shale Gas by Carbon Dioxide at Different Geological Depths. *Chem. Eng. Sci.* **2016**, *156*, 121-127.

20. Sun, H.; Zhao, H.; Qi, N.; Qi, X.; Zhang, K.; Li, Y. Molecular Insight into the Micro-Behaviors of CH<sub>4</sub> and CO<sub>2</sub> in Montmorillonite Slit-Nanopores. *Mol. Simul.* **2017**, *43*, 1004-1011.
21. Santos, M. S.; Franco, L. F.; Castier, M.; Economou, I. G. Molecular Dynamics Simulation of N-Alkanes and CO<sub>2</sub> Confined by Calcite Nanopores. *Energy Fuels* **2018**, *32*, 1934-1941.
22. Wang, R.; Peng, F.; Song, K.; Feng, G.; Guo, Z. Molecular Dynamics Study of Interfacial Properties in CO<sub>2</sub> Enhanced Oil Recovery. *Fluid Phase Equilibr.* **2018**, *467*, 25-32.
23. Pu, J.; Qin, X.; Gou, F.; Fang, W.; Peng, F.; Wang, R.; Guo, Z. Molecular Modeling of CO<sub>2</sub> and N-Octane in Solubility Process and A-Quartz Nanoslit. *Energies* **2018**, *11*, 3045.
24. Le, T.; Striolo, A.; Cole, D. R. CO<sub>2</sub>-C<sub>4</sub>H<sub>10</sub> Mixtures Simulated in Silica Slit Pores: Relation between Structure and Dynamics. *J. Phys. Chem. C* **2015**, *119*, 15274-15284.
25. Sun, H.; Zhao, H.; Qi, N.; Qi, X.; Zhang, K.; Sun, W.; Li, Y. Mechanistic Insight into the Displacement of CH<sub>4</sub> by CO<sub>2</sub> in Calcite Slit Nanopores: The Effect of Competitive Adsorption. *RSC Advances* **2016**, *6*, 104456-104462.
26. Le, T.; Ogbe, S.; Striolo, A.; Cole, D. R. N-Octane Diffusivity Enhancement Via Carbon Dioxide in Silica Slit-Shaped Nanopores—a Molecular Dynamics Simulation. *Mol. Simul.* **2016**, *42*, 745-752.
27. Liu, B.; Shi, J.; Sun, B.; Shen, Y.; Zhang, J.; Chen, X.; Wang, M. Molecular Dynamics Simulation on Volume Swelling of CO<sub>2</sub>-Alkane System. *Fuel* **2015**, *143*, 194-201.
28. Zhang, J.; Pan, Z.; Liu, K.; Burke, N. Molecular Simulation of CO<sub>2</sub> Solubility and Its Effect on Octane Swelling. *Energy Fuels* **2013**, *27*, 2741-2747.
29. Zheng, S.; Li, H. A.; Sun, H.; Yang, D. Determination of Diffusion Coefficient for Alkane Solvent-CO<sub>2</sub> Mixtures in Heavy Oil with Consideration of Swelling Effect. *Ind. Eng. Chem. Res.* **2016**, *55*, 1533-1549.
30. Jin, L.; Hawthorne, S.; Sorensen, J.; Pekot, L.; Kurz, B.; Smith, S.; Heebink, L.; Herdegen, V.; Bosshart, N.; Torres, J. Advancing CO<sub>2</sub> Enhanced Oil Recovery and Storage in Unconventional Oil Play—Experimental Studies on Bakken Shales. *Appl. Energ.* **2017**, *208*, 171-183.
31. Eide, O.; Ersland, G.; Brattekas, B.; Haugen, A.; Graue, A.; Ferno, M. CO<sub>2</sub> EOR by Diffusive Mixing in Fractured Reservoirs. *Petrophysics* **2015**, *56*, 23-31.
32. Lwisa, E. G.; Abdulkhalek, A. R. In *Enhanced Oil Recovery by Nitrogen and Carbon Dioxide Injection Followed by Low Salinity Water Flooding for Tight Carbonate Reservoir: Experimental Approach*, IOP Conference Series: Materials Science and Engineering, IOP Publishing: 2018; p 012009.
33. Eide, Ø.; Fernø, M. A.; Alcorn, Z.; Graue, A. Visualization of Carbon Dioxide Enhanced Oil Recovery by Diffusion in Fractured Chalk. *SPE J.* **2016**, *21*, 112-120.
34. Khan, C.; Amin, R.; Madden, G. Effects of CO<sub>2</sub> and Acid Gas Injection on Enhanced Gas Recovery and Storage. *J. Petrol. Explor. Prod. Technol.* **2013**, *3*, 55-60.
35. Huerta, M. M.; Alvarez, J. M.; Jossy, E.; Forshner, K. In *Use of Acid Gas (CO<sub>2</sub>/H<sub>2</sub>S) for the Cyclic Solvent Injection (Csi) Process for Heavy Oil Reservoirs*, SPE Heavy Oil Conference Canada, Society of Petroleum Engineers: 2012.
36. Chialvo, A. A.; Vlcek, L.; Cole, D. R. Acid Gases in CO<sub>2</sub>-Rich Subsurface Geologic Environments. *Rev. Mineral. Geochem.* **2013**, *77*, 361-398.
37. Badmos, S.; Striolo, A.; Cole, D. R. Aqueous Hydrogen Sulphide in Slit-Shaped Silica Nanopores: Confinement Effects on Solubility, Structural and Dynamical Properties. *J. Phys. Chem. C* **2018**.
38. Rother, G.; Krukowski, E. G.; Wallacher, D.; Grimm, N.; Bodnar, R. J.; Cole, D. R. Pore Size Effects on the Sorption of Supercritical CO<sub>2</sub> in Mesoporous Cpg-10 Silica. *J. Phys. Chem. C* **2011**, *116*, 917-922.
39. Causa, M.; Dovesi, R.; Pisani, C.; Roetti, C. Ab Initio Hartree-Fock Study of the MgO (001) Surface. *Surf. Sci.* **1986**, *175*, 551-560.

40. Scamehorn, C.; Hess, A.; McCarthy, M. Correlation Corrected Periodic Hartree–Fock Study of the Interactions between Water and the (001) Magnesium Oxide Surface. *J. Chem. Phys.* **1993**, *99*, 2786-2795.
41. Ho, T. A.; Striolo, A. Water and Methane in Shale Rocks: Flow Pattern Effects on Fluid Transport and Pore Structure. *AIChE J.* **2015**, *61*, 2993-2999.
42. Teich-McGoldrick, S. L.; Greathouse, J. A.; Cygan, R. T. Molecular Dynamics Simulations of Structural and Mechanical Properties of Muscovite: Pressure and Temperature Effects. *J. Phys. Chem. C* **2012**, *116*, 15099-15107.
43. Wang, J.; Kalinichev, A. G.; Kirkpatrick, R. J.; Cygan, R. T. Structure, Energetics, and Dynamics of Water Adsorbed on the Muscovite (001) Surface: A Molecular Dynamics Simulation. *J. Phys. Chem. B* **2005**, *109*, 15893-15905.
44. White, W. M. *Geochemistry*; John Wiley & Sons, 2013.
45. Tuladhar, A.; Chase, Z. A.; Baer, M. D.; Legg, B. A.; Tao, J.; Zhang, S.; Winkelman, A. D.; Wang, Z.; Mundy, C. J.; De Yoreo, J. J. Direct Observation of the Orientational Anisotropy of Buried Hydroxyl Groups inside Muscovite Mica. *J. Am. Chem. Soc.* **2019**, *141*, 2135-2142.
46. Sakuma, H.; Kawamura, K. Structure and Dynamics of Water on Li<sup>+</sup>, Na<sup>+</sup>, K<sup>+</sup>, Cs<sup>+</sup>, H<sub>3</sub>O<sup>+</sup>-Exchanged Muscovite Surfaces: A Molecular Dynamics Study. *Geochim. Cosmochim. Acta* **2011**, *75*, 63-81.
47. Sakuma, H.; Kawamura, K. Structure and Dynamics of Water on Muscovite Mica Surfaces. *Geochim. Cosmochim. Acta* **2009**, *73*, 4100-4110.
48. Cygan, R. T.; Liang, J.-J.; Kalinichev, A. G. Molecular Models of Hydroxide, Oxyhydroxide, and Clay Phases and the Development of a General Force Field. *J. Phys. Chem. B* **2004**, *108*, 1255-1266.
49. Martin, M. G.; Siepmann, J. I. Transferable Potentials for Phase Equilibria. 1. United-Atom Description of N-Alkanes. *J. Phys. Chem. B* **1998**, *102*, 2569-2577.
50. Kamath, G.; Potoff, J. J. Monte Carlo Predictions for the Phase Behavior of H<sub>2</sub>S+N-Alkane, H<sub>2</sub>S+ CO<sub>2</sub>, CO<sub>2</sub>+ CH<sub>4</sub> and H<sub>2</sub>S+ CO<sub>2</sub>+ CH<sub>4</sub> Mixtures. *Fluid Phase Equilibr.* **2006**, *246*, 71-78.
51. Coasne, B.; Galarneau, A.; Di Renzo, F.; Pellenq, R. Molecular Simulation of Nitrogen Adsorption in Nanoporous Silica. *Langmuir* **2010**, *26*, 10872-10881.
52. Allen, M. P.; Tildesley, D. J. *Computer Simulation of Liquids*; Oxford University Press: Oxford, UK, 2004.
53. Essmann, U.; Perera, L.; Berkowitz, M. L.; Darden, T.; Lee, H.; Pedersen, L. G. A Smooth Particle Mesh Ewald Method. *J. Chem. Phys.* **1995**, *103*, 8577-8593.
54. Van Der Spoel, D.; Lindahl, E.; Hess, B.; Groenhof, G.; Mark, A. E.; Berendsen, H. J. C. Gromacs: Fast, Flexible, and Free. *J. Comput. Chem.* **2005**, *26*, 1701-1718.
55. Abraham, M. J.; Murtola, T.; Schulz, R.; Páll, S.; Smith, J. C.; Hess, B.; Lindahl, E. Gromacs: High Performance Molecular Simulations through Multi-Level Parallelism from Laptops to Supercomputers. *SoftwareX* **2015**, *1*, 19-25.
56. Song, Y.; Dai, L. L. The Shear Viscosities of Common Water Models by Non-Equilibrium Molecular Dynamics Simulations. *Mol. Simul.* **2010**, *36*, 560-567.
57. Sun, H.; Zhao, H.; Qi, N.; Li, Y. Simulation to Enhance Shale Gas Recovery Using Carbon Dioxide in Silica Nanopores with Different Sizes. *Energy Technol.* **2017**, *5*, 2065-2071.
58. Kinaci, A.; Haskins, J. B.; Çağın, T. On Calculation of Thermal Conductivity from Einstein Relation in Equilibrium Molecular Dynamics. *J. Chem. Phys.* **2012**, *137*, 014106.
59. Kadoura, A.; Narayanan Nair, A. K.; Sun, S. Molecular Dynamics Simulations of Carbon Dioxide, Methane, and Their Mixture in Montmorillonite Clay Hydrates. *J. Phys. Chem. C* **2016**, *120*, 12517-12529.





Graphic for the Table of Contents